THE CRYSTAL STRUCTURE OF SELENOUREA

THE CRYSTAL STRUCTURE OF SELENOUREA AT 20°C AND AT -100°C.

Ву

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SCOPE AND CONTENTS: The trigonal form of selenourea has been examined by X-ray crystallographic methods both at room temperature and at a lower temperature ( $-100^{\circ}C$ ). The lattice parameters of earlier workers have been refined and the space group assignment, which requires nine crystallographically distinct SeC(NH<sub>2</sub>)<sub>2</sub> molecules, confirmed. The crystal structure was solved using a novel Patterson technique and differences Fourier syntheses, and was refined by least squares using three-dimensional data obtained at  $-100^{\circ}C$ . Both the molecular dimensions and the crystal structure found are compared to other members of the urea series, and the similarity of the crystal structure to the urea- and thioureahydrocarbon complexes discussed. Evidence is presented of N-H...Se hydrogen bonds, supporting their recent discovery in chemically related compounds.

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#### CHAPTER I

#### INTRODUCTION

The crystal structure of selenourea,  $SeC(NH_2)_2$ , has been studied by single crystal X-ray diffraction techniques at room temperature and  $-100^{\circ}C$ . Through this work, we intended to gain information on the structure of the selenourea molecule, the packing of these simple molecules in the solid state, and as to the existence of N-H...Se hydrogen bonds. The study was also concerned with the possibility of ferroelectricity in this molecular crystal. The structure found will be compared to those of the chemical analogues urea,  $OC(NH_2)_2$ , and thiourea,  $SC(NH_2)_2$ , and to their crystal complexes with hydrocarbons, with particular reference to the packing and degree of hydrogen bonding in these structures.

Selenourea was first propared by Verneuil (1886) by reaction of hydrogen selenide with an aqueous solution of cyanamide,

H<sub>2</sub>Se + H<sub>2</sub>NCN - SeC(NH<sub>2</sub>)<sub>2</sub>,

but until recently few workers have made any consistent study of its physical properties. There is not even agreement in the literature on the melting point, the usual criterion of purity for molecular solids. A number of reported values are:-

Recrystallisation From	Melting Point (°C)	Reference
Water	200 (dec.)	Verneuil (1886)
Water	213	Backer & Bos (1943)
Ethahol	210-211 (dec.)	Mautner (1956)
Water	235 (dec.)	Hope (1964)

The crystal structures of urea (Hendrichs, 1928) and thiourea (Wyckoff and Corey, 1932) were among the earliest of molecular compounds to be solved. The urea structure, which has since been confirmed by both electron (Lobachev and Vainshtein, 1961) and neutron diffraction (Worsham, Levy and Peterson, 1957), contains two molecules in a tetragonal unit cell of space group  $D_{4d}^3-P\bar{4}2m$  with a = 5.661Å, c = 4.712Å. The structure is illustrated in Figure I-1. The oxygen atom of each molecule is linked, by hydrogen bonds of length 3.0Å, to two nitrogen atoms on neighbouring molecules related by the  $\bar{4}$  axis, and also to the two nitrogen atoms of the molecule related by the translations along this axis. This open framework of hydrogen bonds prevents other contacts between the urea molecules. The most recent X-ray studies are due to Sklar, Senko and Post (1961) and Caron and Donchue (1964). The former is of peculiar relevance to the



Figure I-1

The crystal structure of urea projected on (OOl).

present work, in being carried out at two temperatures,  $-140^{\circ}C$ and  $28^{\circ}C$ .

Recent accurate X-ray determinations of the room temperature structure of thiourea have been carried out by Kunchur and Truter (1958), Zvonkova, Krivnov and Khvatkina (1966) and Truter (1967). The conclusions drawn are in agreement with an electron diffraction study by Dvoryankin and Vainshtein (1958). The structure involves four molecules in an orthorhombic cell, of space group  $D_{2h}^{16}$ -Pnma, and is illustrated in Figure I-2. The thiourea molecule has a plane of symmetry passing through the carbon and sulphur atoms, requiring the nitrogen atoms to be equivalent. The nitrogen atoms are found to be coplanar with the carbon and sulphur atoms, although this is not required by the space group symmetry.

Solomon (1957) reported that thiourea was ferroelectric at low temperatures, and Goldsmith and White (1958) showed that this results from small relative displacements of the molecules as a whole in a low temperature phase, stable below  $169^{\circ}$ K, of space group  $C_{2v}^2-P2_1$ ma, where the asymmetric unit is doubled in size. These displacements occur in either of two possible ways, which result in diametrically opposite directions for the residual dipole moment in this non-centrosymmetric structure. Since these displacements from the high temperature structure are small, the two forms have



Figure I-2

The crystal structure of thiourea projected on (001).

only a small barrier to interconversion, which can be overcome by the application of an electric field of the correct polarity. Such an electric field will convert the crystal into the form with reversed moment; this phenomenon is called ferroelectric 'switching'. Again this low temperature structure has been confirmed by electron diffraction (Dvoryankin and Vainshtein, 1961) which in addition reveals the hydrogen bonding characteristics.

Kitaigorodskii (1961, 1965) is of the opinion that the determining factor in the crystal structure of organic compounds is steric, that is, the intermolecular interactions can be approximated as those arising from non-interpenetrable molecules. This 'hard molecule' model is in fact a good approximation from the point of view of lattice energy, since intermolecular forces can be shown to be not significantly affected by crystal fields in molecular crystals. Some examples substantiating this will be presented in the Discussion. The minimum free energy of the crystal then arises from the highest packing density, since the lattice energy is determined by the number of intermolecular contacts giving rise to the non-directional dispersion forces, commonly called van der Waals forces. What Kitaigorodskii has done is to derive a list of space groups which are likely to result in high packing density, through consideration of the effects of the various symmetry elements involved on a molecule of arbitrary shape.

These predictions are well borne out in practice. Dipolar

6.

interactions have little effect on this scheme, since they appear only to single out one particular free energy minimum from several available on purely steric grounds. The same is usually true of hydrogen bonds, despite their directed nature and the fact that they are found in practice to be saturated, that is, all available groups N-H and O-H are used in hydrogen bonding. However deviations from close-packing, due to the directional requirements of strong hydrogen bonds, do sometimes occur; this is well illustrated by comparing the structures of urea and thiourea.

With regard to urea, Kitaigorodskii points out that the structure is relatively 'loose' compared with the packing in most organic crystals, and attributes this to the steric requirements of the large number of hydrogen bonds involved. He also shows that thiourea would suffer from further large losses in packing density if it assumed the urea structure; thus the structure found for thiourea is of lower symmetry, but higher packing density, and is one of those theoretically predicted by Kitaigorodskii for the close-packing of organic molecules.

Urea and thiourea form inclusion compounds with certain hydrocarbons. The structure of the crystalline compounds formed by urea with straight-chain aliphatic hydrocarbons was elucidated by Smith (1952). The space group is  $C6_1^2$  or the enantiomorphous  $C6_5^2$ . The unit cell contains six urea molecules, the lattice

constants being a = 8.230(4) Å, <sup>\*</sup>c = 11.005(5)Å. These urea molecules form a honeycomb structure by hydrogen bonding among themselves, leaving six-sided channels which contain the hydrocarbon molecules. The packing density in these crystals is greater than for either component alone which is further evidence in favour of Kitaigorodskii's principles.

A completely analagous phenomenon was also found for thiourea; in this case the structure determination is due to Lenné (1954). The basic structure has a similar honeycomb form, but the symmetry is reduced, the space group now being the rhombohedral  $R\overline{3}c$ . The unit cell has the dimensions a = 15.8Å, c = 12.5Å, contains 18 molecules of thiourea, and involves three six-sided channels, which are now large enough to contain cyclic hydrocarbons such as cyclohexane.

Recently three accurate crystal structure determinations of substituted selenoureas have been completed, phenylbenzoylselenourea (I) by Hope (1965), 2-selenouracil (II) by Tsernoglou (1966) and 2:4-diselenouracil (III) by Shefter, James and Mautner (1966).

Throughout this thesis errors (standard deviations where least squares is involved) will be designated by including the error in the final digit(s) in parentheses, e.g. 8.230(4)Å is equivalent to 8.230±0.004Å.



All of these studies found the selenourea moiety, save possibly for the hydrogen atoms, to have the planar form found for the other members of the series, with Se-C-N and N-C-N angles near 120°. Some interatomic distances found are listed in Table I-lA. It will be shown below that these are in keeping with predictions based on the other analogues.

These three studies further produced all the crystallographic evidence to date for hydrogen bonds N-H...Se. The lengths of these hydrogen bonds are also listed in Table I-1A. For compounds II and III the interpretation is strengthened by the determination of the corresponding sulphur compounds, 2-thiouracil (Tsernoglou, 1966) and 2:4-dithiouracil (Shefter and Mautner, 1967). For example the N-H...Se distance in 2-thiouracil is 3.32Å, close

## TABLE I-1

Some bond lengths and angles in compounds related to selenourea.

Atoms.	Distance (Å).	Compounds.
C-Se	1.81 1.84	I II
C-N	1.32 1.40 1.34 1.34	I II
N-HSe	3.83 3.42 3.47 3.75	I II III III

## A. Substituted selenourcas.

## B. Urea and thiourea.

	C-X(A)	C-N(R)	$\underline{X-C-N(^{\circ})}$	N-C-N(°)	Reference.
X=0	1.262(3)	1.341(3)	120.9(2)	118.2(2)	Sklar,Senko & Post (1961)
	1.276(8)	1.356(7)	120.5(6)	119.0(3)	Caron and Donohue (1964)
X=S	1.75	1.34	121	118 )	Caldemith and White (1050)
	1.73	1.34	121	118	Goldsmith and white (1999)
	1.720(9)	1.340(6)	120.5(5)	119.0(5)	Truter (1967)

to the value of 3.30Å found in trimethylenethiourea (Diaz and Truter, 1964). For all three compounds two molecules are linked across a centre of symmetry by these hydrogen bonds, to form the grouping

> C Se... H - N

The existence of similarly hydrogen bonded dimers has been deduced for 2-pyridselenone (IV) in solution by Krackov, Lee and Mautner (1965) on the basis of dipole moment and molecular weight measurements.

The urea series XC(NH<sub>2</sub>)<sub>2</sub> have been the subject of a large amount of interest, as far as accurate molecular dimensions are concerned, since the system appears to exhibit electronic delocalisation, which increases the bond order of the C-N bonds and decreases the bond order of the C-X bond. This delocalisation also leads to hydrogen positions coplanar with the rest of the molecule. Typical bond lengths and angles found in this series are given in Table I-1B.

The first solid state study reported for selenourea was that of Mikhailov, Lundin, Gabuda and Aleksandrov (1961), who applied nuclear magnetic resonance techniques to powdered samples. They interpreted their results, on the basis of a similar molecular structure to thiourea, as indicating a distance of 1.75Å between protons on the same NH<sub>2</sub> group. This is similar to urea and thiourea. They also found that there is intermittent rotation of the molecule about the C-Se axis at temperatures above 130°K, and, by consideration of the variation of the second moment with temperature, showed that the barrier to the rotation is around 6 kilocalories per mole, which can be compared with values of 12.7 kilocalories per mole for urea and 9 kilocalories per mole for thiourea, both determined by the same technique. It is unfortunate that there is no crystal data on the samples used by Mikhailov et al., in light of the subsequent reports by other workers of two separate crystalline modifications of selenourea.

Dvoryankin and Ruchkin (1962) reported an initial electron diffraction study of selenourea. The crystals, prepared by recrystallisation from ethanol-water solutions, were found to belong to the space group Pnma, with the lattice constants here compared to the high temperature form of thiourea, which belongs to the same space group.

	a(Å)	b(Å)	c(A)	$v(A^2)$	Z
Selenourea	6.48	8.75	7.04	399	4
Thiourea	7,65	8.53	5.52	361	4

This phase has not been confirmed by any of the subsequent

X-ray .work.

Preliminary accounts of the unit cell dimensions and contents of a second modification have been given by Kondrashev and Andreeva (1963), Perez Rodriguez, Cubero and Lopez-Castro (1964), and Hope (1965), all using X-ray methods. Kondrashev and Andreeva studied single crystals grown from aqueous solution, and assigned them to one of the enantiomorphous pair of space groups  $P_{j_1}$ ,  $P_{j_2}$ , on the basis of the systematic absences (00 $\ell$ ,  $\ell \neq 3n$ ) found in the diffraction pattern. Perez Rodriguez et al. found the crystal to have the same space group, but noted that careful examination of the intensities is required to eliminate the higher symmetry point groups D3, C3v and D3d. Hope did not venture to assign a space group in his report. The unit cell contains 27 molecules, and is very similar in dimensions to those of the ureaand thiourea- hydrocarbon inclusion compounds which contain 18 XC(NH2)2 molecules. The urea compound requires a transformation to a triply-primitive cell for comparison.

and the second	$\underline{a(A)}$	<u>c(Å)</u>
Selenourea (Perez Rodriguez et al.)	15.34	12.99
Thiourea/cyclohexane	15.8	12.5
Urea/n-hexadecane	14.3	11.0

This suggests that perhaps 18 molecules per unit cell are

involved in a similar honeycomb structure, while the remaining nine are distributed in the channels, three to each. Kondrashev and Andreeva further report that this is born out by the hkO Patterson function, which illustrates the position of the 18 selenium atoms of the honeycomb structure.

The study reported here was commenced in 1962. As the work described above became available, it became clear not only that there was, for various reasons, a great deal of interest in the crystal structure of selenourea, but that some duplication of effort might ensue as a result. In particular, Lopez-Castro (1965) confirmed her intention of carrying out a complete room temperature study. At this point it was decided to conduct a parallel low temperature structure determination; -100°C was chosen as a convenient temperature considering the apparatus available.

The rationale for low temperature crystal structure determination is the following. The diffraction experiment does not record an atom in its equilibrium position, but a distribution of atoms displaced from their equilibrium positions by thermal motion. In fact the variation in intensity with temperature for a monatomic isotropic crystal involved the factor  $e^{-2B(\sin\theta/\lambda)^2}$ , where  $B = 8\pi^2\mu^2$ , and  $\mu^2$  is the mean square displacement of the atom from its equilibrium position.  $\mu^2$  increases with increasing temperature, the variation being close to linear except at very low temperatures (Lonsdale and El Sayed, 1965), and makes it more

difficult to determine the true equilibrium position, and hence chemically important bond lengths. One way of looking at this effect is to note that the intensities decrease with increasing temperature, especially for large values of  $(\sin\theta/\lambda)$ , this results in a smearing out of atoms in electron density maps, because of the decrease in magnitude of these higher order Fourier coefficients. This has been illustrated by Burbank (1953), using the expressions and notation of Cruikshank (1949). The standard deviation  $\sigma(x)$  in a co-ordinate derived from such clectron density maps, is given by

$$\tau(\mathbf{x}) = \frac{\sigma(\mathbf{A_n})}{\mathbf{A_{nn}}}$$
(I-1)

where

and

$$\sigma(A_n) = \frac{2\pi}{aV} \left\{ \sum_{hk \ell} h^2 (\Delta F_{hk \ell})^2 \right\}^{1/2}$$

$$A_{nn} = -\frac{\partial^2 \rho}{\partial x^2}$$

 $\mathbf{C}(\mathbf{A}_{n})$  is simply proportional to the root mean square difference between the observed and calculated structure factors,  $\Delta \mathbf{F}_{hkl}$ , properly weighted with respect to the co-ordinate x. Thus it depends primarily on the accuracy of the data.  $\mathbf{A}_{nn}$ , however, is the curvature of the electron density at the peak position, and Burbank calculated this for various elements and values of B. For example, for carbon

B	0	2	- 4-
A <sub>nn</sub> (electrons/A <sup>5</sup> )	-264	-119	-58

The latter two values of B correspond roughly to  $150^{\circ}$ K and  $300^{\circ}$ K for many molecular compounds; thus we should hope to obtain more accurate atomic positions, and hence interatomic distances, by a factor of about 2 from work at  $-100^{\circ}$ C.

Cruickshank (1960) himself has discussed the accuracy required in intensity measurement to gain sufficient accuracy in interatomic distances, and stresses the importance of low temperature work, suggesting that all molecular Crystal work should be carried out at  $100^{\circ}$ K (-173°C) or lower. In addition to the greater resolution described above, he points out that errors can arise in the apparent equilibrium positions at high temperatures, caused by both molecular libration and anharmonicity in the lattice vibrations. It is difficult to correct for these effects without knowing the detailed nature of the crystal vibrations. This plea has gone unheeded since in practice a very small proportion of crystal structures are determined at low temperatures.

#### CHAPTER II

#### EXPERIMENTAL

#### General remarks.

X-ray diffraction is by now a very well established technique for the determination of the atomic arrangements in crystals. The majority of experimental apparatus and techniques, and of theoretical results, used in this study have been known for several decades, are in worldwide everyday use, and are well described in a number of standard texts (Buerger, 1942, 1960, Lipson and Cochran, 1967, etc.) I therefore feel it unnecessary to attempt a prolonged description of such methods here, and prefer to limit more detailed description to techniques which are either new or unusual.

#### The low temperature apparatus.

The low temperature work used the Nonius Weissenberg camera with low temperature attachment. This consists of an insulated tube, which passes down within, and concentric with, the layer line screen to a position just short of the crystal; a controlled stream of cold nitrogen is passed through this tube and over the crystal. The nitrogen then passes out again through the open end of the layer line screen. The temperature is monitored by means of a copper-constantan thermocouple at the end of the tube nearest the crystal. To maintain the film at room temperature and to prevent frosting, the tube is surrounded by a heating coil, and the layer line slit is scaled with cellophane tape. With these precautions the cooling of the actual film holder amounted to only a few degrees, sufficient to cause condensation only on the most humid days. The cold nitrogen stream is produced in our apparatus by boiling liquid nitrogen by means of electrical heating. Temperatures in the range -150°C to room temperature can be readily achieved and maintained with this apparatus, although the lowest temperatures need careful monitoring and a prodigal expenditure of liquid nitrogen.

#### The determination of lattice parameters.

On crystallisation from water, selenourea forms trigonal prisms, the prominent forms being {OOl}, {OOI} and {IOO}. The identity of the last named prism faces was found by X-ray diffraction, which also showed the Laue symmetry  $C_{3i}$ . Since the only systematic absences observed are OO $\ell$ ,  $\ell \neq 3n$ , this leads immediately to the enantiomorphous pair of space groups P3<sub>1</sub> and P3<sub>2</sub> ( $C_3^2$  and  $C_3^3$ ). No space group with these extinctions occurs

in the centrosymmetric point group Cz;.

The Bragg equation

$$n\lambda = 2d \sin \Theta$$
 (II-1)

relates the spacing d<sub>hk2</sub> between succesive planes of Miller indices hk2 in the crystal lattice. We can therefore, knowing the wavelength of the radiation used, measure the dimensions of the unit cell.

The lattice parameters of selenourea were measured at  $20^{\circ}$ C by superimposing the diffraction pattern of a single crystal of rutile (the tetragonal modification of TiO<sub>2</sub>) on an hO& Weissenberg photograph for selenourea, in both cases using CoK<sup>CC</sup> radiation.<sup>\*</sup> The lattice parameters of rutile have been accurately determined as a = 4.5929(5)Å, c = 2.9591(3)Å by Cromer and

The x-radiations used in this study, with their wavelengths, as listed in International Tables for X-ray Crystallography, and corresponding KB filters, were :-

Radiation	~	d'2	Filter
CoKa	1.78892	1.79278	Fe
CuKX	1.54051	1.54433	Ni
MoKX	0.70926	0.713543	Zr

Herrington (1955). These values were used to correct the selenourea measurements for camera radius and film shrinkage; the rutile measurements were all consistent with an effective camera radius of 2.848 cm. The corrected values of the angles,  $\Theta$ , for 12 hO& reflections were used to determine the lattice parameters by least squares, using the values of Perez Rodriguez et al. as a starting point, and the FORTRAN program DESLS (Robertson,1966) on the I.B.M. 7040 computer, which was used for all computations. The observed angles are tabulated (Table II-IA), together with those calculated from the lattice parameters found by least squares, namely a = 15.285(5)Å, c = 13.007(5)Å at 20°C.

These values correspond to a unit cell whose volume is  $2631\text{A}^3$ . If we assume that the cell contains 27 molecules of the formula  $\text{SeC(NH}_2)_2$ , (formula weight 123.01), then the individual molecular volume is  $97.4\text{A}^3$ , and the calculated density is 2.09. Literature values are 2.06, 2.09 (Kondrashev and Andreeva), and 2.1 (Perez Rodriguez et al.). Since the space group P3<sub>1</sub> contains only 3-fold positions we have further reason to rule out other possibilities. Thus we have nine crystallographically distinct molecules in the unit cell.

If the diffraction patterns at two temperatures of a single crystal are superimposed on one film, and provided the structure and orientation are preserved in going from one temperature to the other, it is possible to determine with a very

### TABLE II-1.

Observed and calculated diffraction angles in the lattice parameter determinations for selenourea.

hk-C	⊖(Obs.) degrees.	e(Calc.) degrees.
300 003 303 600 006 603 306 307 900 009	11.72 11.92 16.83 23.96 24.44 27.10 27.40 31.49 37.50 38.24	11.70 11.91 16.82 23.94 24.39 27.08 27.08 27.39 31.51 37.49 38.27 20.04
903	39.94	39.99

A. Room temperature.

### B. Low temperature.

hk <b>@(*)</b>	$\Delta \Theta$ (Obs.) degrees.	$\Delta \theta$ (Calc.) degrees.
3,3,14	0.48	0.49
6,6,11	0.58	0.56
1,1,15	0.48	0.51
990	0.70	0.70
991	0.65	0.71
2,2,15	0.58	0.57
5,5,13	0.70	0.65
7,7,10	0.68	0.71
993	0.80	0.79
888	0.80	0.82
994	0.90	0.94

(\*) In order of increasing <del>0</del>.

high degree of accuracy the changes in lattice parameters over the temperature range in question. This arises because this relative measurement of the Bragg angle not only automatically corrects for the effects of film shrinkage and camera radius mentioned above, but also, since the same crystal is used throughout, for errors resulting from crystal eccentricity and absorption.

This technique was applied to selenourea, using the Nonius Weissenberg camera. The crystal was oriented so that the hh& reciprocal lattice zone was recorded, and exposures were taken at 20°C and -100°C using CuKX radiation with the camera displaced slightly laterally between exposures. The relative spacing of the  $\alpha_1, \alpha_2$  doublets at both temperatures were measured for 11 reflections, and these were used to determine the change in lattice parameters again using the DESLS program. These were found to be 0.084(1) A in a and 0.057(1) A in c, corresponding to coefficients of thermal expansion averaged over this temperature range of  $\alpha_a = 46 \times 10^{-6}/^{\circ}$ C and  $\alpha_c = 36 \times 10^{-6}/^{\circ}$ C. Since the · changes are known much more accurately than the parameters themselves, we can say that the unit cell dimensions at -100°C are a = 15.201(5)Å, c = 12.950(5)Å. The observed and calculated changes in angle, averaged for each doublet, are tabulated in Table II-1B.

#### The measurement of intensity data.

All intensity data were measured by photographic techniques, using single crystal cameras based on the Weissenberg (1924) or Buerger precession (1944) motions. The measured intensities were assigned standard deviations assuming a constant relative error of 20%, or an error equal to the minimum observable intensity on that particular set of films, whichever was greater. These standard deviations were later used in weighting observations in least squares refinements. As a weighting scheme, this is as simple to apply as that suggested by Hughes (1941), is almost identical in effect, and has the added . advantages of taking account of local variations in Fmin, and of allowing weighted averaging of equivalent structure factors. Absorption corrections were then made as detailed below, and Lorentz and polarisation corrections were applied using the FORTRAN programs WEILPC (Weissenberg) and PRELPC (precession) written in this laboratory. All data were finally converted to structure factors with each layer scaled independently.

The data used for the room temperature study consisted of 3 reciprocal lattice zones, each gathered by a different photographic technique. All other data collected at room temperature were found to be useless for intensity measurement because of the use of twinned crystals. The nature of the twinning will be

described in the Discussion.

The hold data were collected on the Supper Weissenberg camera, using Cu-KQ radiation, and the multiple film technique (Robertson, 1943) with Ilford Ilfex X-ray film (transmission for CuKQ radiation 1/2.7). Since the crystal used was extremely small (less than  $10^{-7}$ c.c.) no absorption correction was made to this data. The intensity measurements for the hkO zone used a somewhat larger crystal, yet still small enough to avoid twinning effects. This time the precession camera was used, along with MoKQ radiation, for which the absorption factor for selenourea is 100cm.<sup>-1</sup>. On the precession camera timed exposures were used rather than the multiple film technique. Approximate absorption corrections were applied to this data by assuming the crystal to be a sphere of radius 0.02cm.( $\mu$ R = 2) and interpolating between the values in the table in International Tables for X-ray Crystallography.

For the hh<sup>2</sup> data a larger crystal could be used, since for this projection the twinning does not interfere. This allowed the use of the integrating precession camera (Nordman, Patterson, Weldon and Supper, 1955), and the measurement of the intensities with a Leeds and Northrup Gl microdensitometer. The crystal used was a thin plate, oriented so that the surface of the plate was parallel to the film. Again a series of timed exposures were taken. No absorption correction was applied to this data.

The low temperature data were all taken on the Nonius

Weissenberg camera, using CuK $\propto$  radiation and equi-inclination geometry (Buerger 1942). The crystals used were all primes of volume less than 5 X  $10^{-7}$ c.c., since it had been found during the room temperature study that crystals of volume  $10^{-5}$ c.c. or greater were frequently twinned. The small crystals made the exposure time required for integrated photographs inconveniently long, considering the temperature had to be monitored at about hourly intervals, and the duration of a 25 litre can of liquid nitrogen was 10 - 12 hours. As it was, the low temperature study was accomplished entirely by non-integrated multiple film exposures, and exposure times of 10 - 12 hours were barely adequate. The dimensions of the crystals used (length and mean width) and the layers collected with each are listed in Table II-2.

It was found convenient to index the h,h+n, & data on the basis of the orthohexagonal cell, and to compute the absorption, Lorentz and polarisation corrections in this form, before transforming to the primitive cell.

The data were corrected for absorption by assuming that the crystals were cylinders (hkn data) or rectangular prisms (hn $\ell$  and h,h+n, $\ell$  data) of the dimensions stated, and ignoring edge effects. In this approximation the path length to and from any point in the crystal for a reflection of radial co-ordinate  $\mathbf{s}^*$  (which equals

This co-ordinate system and nomenclature is that described in Buerger (1942).

# TABLE II-2.

Details of the layer line data used in the low temperature study.

Layer	Crystal Dimensions (cm.XlO <sup>3</sup> )		Number of Reflections		Weighted R	
Indices	Length	Diameter	Observed	Unobserved	Total	Factor on Merging.
			1. M. 1.	2 Section 1	1.0	B. C. C. C. C.
hkO	15	3	107	57	164	0.178
hkl	20	4	85	243	328	0.209
hk2	20	2	97	231	328	0.088
hk3	20	2	129	97	226	0.128
hOL	18	5	143	275	418	0.171
hlł	.18	5	146	532	678	0.208
h2 <b>£</b>	18	5	340	151	491	0.160
hh <b>l</b>	- 20	4	130	100	230	0.171
h, h+1, <b>2</b>	20	4	289	149	438	0.167
h,h+2, <b>e</b>	20	4	336	113	449	0.157
Total			1802	1948	3750	0.159

2 sin  $\Theta$  for zero layer) and angular co-ordinate  $\overline{\mathbf{Q}}$ , measured at equi-inclination angle  $\mathbf{v}$ , is simply sec  $\mathbf{v}$  times the path length for the zero level reflection of co-ordinates  $\mathbf{s}$  and  $\overline{\mathbf{Q}}$ . The cylindrical absorption corrections were interpolated from the values given in International Tables for X-ray Crystallography, using  $\mu$ R sec  $\mathbf{v}$  where appropriate. The correction for the rectangular prism was made using the Albrecht (1939) method, programmed on the I.B.M. 7040 computer. The program computed for each reflection the attenuation of the beam diffracted by 100 points (distributed on a regular 10 X 10 grid) within the crystal, and corrected the observed intensity by multiplying by the inverse of the mean attenuation.

#### The solution of the structure.

The intensity of diffraction for any particular reflection, with which we can associate the reciprocal lattice point  $\underline{H}$ , for which the momentum transfer on scattering is given by the Laue equation

$$\mathbf{k} - \mathbf{k} = 2 \mathbf{\Pi} \mathbf{H} \tag{II-2}$$

is proportional to the square of the structure factor  $F_{\underline{H}}$ . The structure factor is related to the electron density  $\rho$  at a point

r within the unit cell by the expression

$$F_{\underline{H}} = \int \rho(\underline{r}) e^{2\pi i \underline{H} \cdot \underline{r}} d\underline{r} \qquad (II-3)$$

If we assume the electron density to be composed of N spherically symmetric atoms, and, ignoring temperature effects for the moment, we have

$$\mathbf{F}_{\underline{H}} = \sum_{j=1}^{N} \mathbf{f}_{j}(|\underline{H}|) e^{2\pi \underline{i}\underline{H}} \cdot \underline{r}_{j} \qquad (II-4)$$

where  $f_j$  is called the atomic scattering factor. The structure factor and the electron density are the Fourier transforms of each other, that is we can evaluate the electron density, knowing the  $F_H$ 's, as the Fourier series

$$\boldsymbol{\rho}(\underline{\mathbf{r}}) = \frac{1}{\overline{\mathbf{v}}} \sum_{\mathrm{H}} F_{\mathrm{H}} e^{-2\pi i \underline{\mathrm{H}} \cdot \underline{\mathbf{r}}}$$
(II-5)

If we do not know the phases of the complex quantities  $F_{\underline{H}}$ , we can construct another series, the Patterson function

$$P(\underline{u}) = \frac{1}{\overline{v}} \sum_{\underline{H}} F_{\underline{H}} F_{\underline{H}} F_{\underline{H}}^{*} e^{-2\pi i \underline{H} \cdot \underline{u}}$$
$$= \frac{1}{\overline{v}} \sum_{\underline{H}} \sum_{j=1}^{N} \sum_{k=1}^{N} f_{j} f_{k} e^{2\pi i \underline{H} \cdot (\underline{r}_{k} - \underline{r}_{j})} e^{-2\pi i \underline{H} \cdot \underline{u}} \quad (II-6)$$
This function then contains  $N^2$  peaks, of magnitude  $f_j f_k$ , located at  $\underline{r}_j - \underline{r}_k$  from the origin. If we can interpret the Patterson in terms of such a vector set (Buerger, 1959) we have a solution for the crystal structure. We can then refine the parameters in our model to improve the agreement between our observed and calculated structure factors.

When we apply this technique to selenourea, we find that we can ignore interactions other than Se-Se, so that we have  $27^2$ vectors to locate and identify. We can reduce this number by ignoring origin peaks j=k, taking account of the centre of symmetry  $(\underline{r_k}-\underline{r_j}$  must exist as well as  $\underline{r_j}-\underline{r_k}$ ) and the 3-fold axis to 117 distinct peaks. Even then, this is a problem somewhat beyond the brute force approach.

However, a marked pattern of more intense reflections was noted on the films corresponding to the indices -h+k = 3n. In the ho**2** zone, for example, the ratio of the most intense reflections of each class,  $I_{003}/I_{103}$ , is about 9. For the hkO zone it is even higher. This implies a marked pseudosymmetry, a hexagonal subcell of dimensions one third of the true cell, having the same c-axis and the orientation shown in Figure II-1. This rotation about <u>c</u> involves the transformation matrix

$$\mathbf{T} = \begin{pmatrix} 1/3 & -1/3 & 0 \\ 1/3 & 2/3 & 0 \\ 0 & 0 & 1 \end{pmatrix}$$
(II-7)



### Figure II-1

The relationship in orientation and dimensions between the small pseudo-cell and the true unit cell.

where T is here quoted in the form with determinant 1/3, corresponding to the 3-fold decrease in cell volume. The transpose of the inverse matrix is

$$R = \begin{pmatrix} 2 & -1 & 0 \\ 1 & 1 & 0 \\ 0 & 0 & 1 \end{pmatrix}$$

This governs the corresponding reciprocal lattice transformation, so that we have the relationships

$$A = Ta X = Rx$$
$$B = Rb H = Th$$

where a,b,x, and h are the real axes, reciprocal axes, real space co-ordinates and reciprocal space co-ordinates respectively for the superlattice, and their upper case equivalents the corresponding sublattice quantities.

Further, the layers hkO, hkl and hk2 (although the latter two were not used for intensity measurements) showed a further . pattern of intense spots, as well as the rule described above. These correspond to

-H+K+L = 3n

where HKL are the indices in terms of the small cell.

This implies that within the subcell there is an approximate rhombohedral centering, that is, there are equivalent atoms at the points (0,0,0), (2/3,1/3,1/3) and (1/3,2/3,2/3) in terms of the subcell. Applying

$$x = R^{-1}X$$

these become (0,0,0), (1/3,0,1/3) and (2/3,0,2/3).

Taken together with the translations corresponding to the subcell, we have the nine approximately equivalent positions

(0,0,0)	(1/3, 0, 1/3)	(2/3, 0, 2/3)
(2/3,1/3,0)	(0, 1/3,1/3)	(1/3,1/3,2/3)
(1/3,2/3,0)	(2/3,1/3,1/3)	(0, 2/3, 2/3)

We can refine this approach still further, by attempting to solve the Patterson function within the small cell. This approach is justified in the following way. We assume that the superlattice involves a set of small displacements from an ideal sublattice structure. We then have, for an M-fold increase in the unit cell

$$\mathbf{F}_{\underline{\mathbf{H}}} = \sum_{j=1}^{N} \sum_{\boldsymbol{\ell}=1}^{\underline{M}} \mathbf{f}_{j} e^{2\pi i \underline{\mathbf{H}} \cdot (\underline{\mathbf{r}}_{j} + \underline{\boldsymbol{\delta}}_{j\boldsymbol{\ell}} + \underline{\boldsymbol{\delta}}_{\boldsymbol{\ell}})}$$

where  $r_j$  is the average position in the ideal cell of atom j, and

 $\underline{S}_{j\ell}$  is its displacement from this position in the subcell  $\underline{L}$ , which we always assume small, and  $\underline{\Delta} \underline{\ell}$  is the vector from the origin to the origin of subcell  $\underline{L}$ , that is, it is a sublattice translation. For sublattice reflections, we have

$$F_{\underline{H}} = \sum_{j=1}^{N} \sum_{\ell=1}^{M} f_{j} e^{2\pi i \underline{H} \cdot \underline{r}_{j}} e^{2\pi i \underline{H} \cdot \underline{\delta}_{j} \ell}$$

$$= \sum_{j=1}^{N} \sum_{\ell=1}^{M} f_{j} e^{2\pi i \underline{H} \cdot \underline{r}_{j}} (1 + 2\pi i \underline{H} \cdot \underline{\delta}_{j} \ell - \frac{4\pi^{2}}{2!} (\underline{H} \cdot \underline{\delta}_{j} \ell)^{2} \dots)$$

$$\approx \sum_{j=1}^{N} \sum_{\ell=1}^{M} f_{j} e^{2\pi i \underline{H} \cdot \underline{r}_{j}} (1 - 2\pi^{2} (\underline{H} \cdot \underline{\delta}_{j} \ell)^{2}) \quad (II - 9)$$

This implies that using the sublattice reflections, we can construct a Patterson function which we can attempt to solve for the ideal structure, since this approximates, especially for small  $\underline{H}$ 

$$M \sum_{j=1}^{N} f_{j} e^{2\pi i \underline{H} \cdot \underline{r}_{j}}$$

To illustrate what sort of approximation this is, I shall introduce a special case, where only a doubling of the unit cell is involved. Then  $\underline{\delta}_{j}$ , =  $\underline{\delta}_{j1}$  =  $-\underline{\delta}_{j2}$  and

$$F_{\underline{H}} = 2 \sum_{j=1}^{N} f_{j} e^{2\pi i \underline{H} \cdot \underline{r}_{j}} \cos 2\pi (\underline{H} \cdot \underline{\delta}_{j})$$

For the superlattice reflections, we have

$$F_{\underline{H}} = \sum_{j=1}^{N} \sum_{\ell=1}^{M} f_{j} e^{2\pi i \underline{H}} \cdot (\underline{r}_{j} + \Delta \ell) (1 + 2\pi i \underline{H} \cdot \underline{\delta}_{j} \ell^{-} \frac{4\pi^{2}}{2!} (\underline{H} \cdot \underline{\delta}_{j} \ell)^{2} \dots)$$

$$= \sum_{j=1}^{N} \sum_{\ell=1}^{M} f_{j} e^{2\pi i \underline{H}} \cdot (\underline{r}_{j} + \Delta \ell) (2\pi i \underline{H} \cdot \underline{\delta}_{j} \ell^{-} \frac{4\pi^{2}}{2!} (\underline{H} \cdot \underline{\delta}_{j} \ell)^{2} \dots)$$

$$\approx \sum_{j=1}^{N} \sum_{\ell=1}^{M} 2\pi i (\underline{H} \cdot \underline{\delta}_{j} \ell) f_{j} e^{2\pi i \underline{H}} \cdot (\underline{r}_{j} + \Delta \ell) (2\pi i \underline{H} \cdot \underline{\delta}_{j} \ell) (2\pi i \underline{H} \cdot \underline{\delta}_{$$

For the special case above

$$F_{\underline{H}} = \sum_{j=1}^{N} f_{j} e^{2\pi i \underline{H} \cdot \underline{r}_{j}} \left\{ e^{2\pi i \underline{H} \cdot \underline{\delta}_{j}} - e^{-2\pi i \underline{H} \cdot \underline{\delta}_{j}} \right\}$$
$$= \sum_{j=1}^{N} 2i f_{j} e^{2\pi i \underline{H} \cdot \underline{r}_{j}} \sin 2\pi (\underline{H} \cdot \underline{\delta}_{j})$$

Now, if we construct a Patterson using only the superlattice reflections we get

$$P(\underline{u}) = \frac{1}{\mathbf{v}} \sum_{\underline{H}}^{\mathbf{w}} F_{\underline{H}} F_{\underline{H}}^{*} e^{-2\pi i\underline{H} \cdot \underline{u}}$$

$$= \frac{1}{\mathbf{v}} \sum_{\underline{H}}^{\mathbf{w}} \sum_{j=1}^{\mathbf{w}} \sum_{k=1}^{\mathbf{w}} \sum_{\underline{e}=1}^{\mathbf{w}} \sum_{m=1}^{\mathbf{w}} 4\pi^{2} (\underline{H} \cdot \underline{\delta}_{je}) (\underline{H} \cdot \underline{\delta}_{km}) f_{j} f_{k} e^{2\pi i\underline{H} \cdot (\underline{r}_{k} + \underline{\Delta}_{m} - \underline{r}_{j} - \underline{\Delta}_{e})}$$

$$e^{-2\pi i\underline{H} \cdot \underline{u}} \qquad (II-11)$$

which can be interpreted as a function with features at the points  $\underline{\mathbf{r}}_{k} - \underline{\mathbf{r}}_{j} + \underline{\Delta}_{m} - \underline{\Delta}_{\boldsymbol{\ell}}$ ,  $(\underline{\Delta}_{m} - \underline{\Delta}_{\boldsymbol{\ell}})$  being a subcell translation) whose

magnitudes are proportional to  $(\underline{H}, f_{j} \underline{\delta}_{j} \underline{\ell}) (\underline{H}, f_{k} \underline{\delta}_{km})$  where the bar represents an average over <u>H</u>. The features on the normal Patterson at these points are of magnitude  $f_{j}f_{k}$ , so we can now find the value of  $(\underline{H}, f_{j} \underline{\delta}_{j} \underline{\ell}) (\underline{H}, f_{k} \underline{\delta}_{km})$ . This gives us the sign of the displacements of atoms j $\underline{\ell}$  and km relative to each other, from which we may hope to solve the superlattice structure. By the way, the height of the origin peak here will be

$$\frac{1}{V} \sum_{j \neq e} 4\pi^2 (\underline{\mathbb{H}}_{j \neq j e})^2$$

that is, proportional to the weighted mean square displacement of all the atoms from the average structure.

The interpretation of the Patterson function for structures involving superlattices, using the two sets of reflections independently, has been suggested by Buerger (1959). His analysis, however, is unsuitable for the present case, since it is in terms of a 'complement structure' which consists of electron density added to the substructure. Mackay (1953) was the first to discuss superstructures in terms of displacements from an average structure; he found that the ratio between the average intensities of the two classes of reflections is given, for equal atoms,

$$R = 4\pi^2 (\underline{H} \cdot \underline{\delta}_j)^{2(*)}$$

(\*) A more accurate result is  $4\pi^2 (\underline{H} \cdot \underline{\delta}_j)^2 / [1 - 4\pi^2 (\underline{H} \cdot \underline{\delta}_j)^2]$ 

or assuming random displacements

$$R = 2\pi^2 d^{*2} \delta^2$$
 (II-12)

where **6** is the root mean square displacement, d the magnitude of the reciprocal lattice vector <u>H</u>. This is equivalent to the result above for the height of the origin peak. This analysis allowed Mackay to calculate the magnitude of the displacements involved in the structure of  $Ca_3(PO_4)_2$ , where M is 2.

The 'M=2' problem was carried further by Sakurai (1958), who introduced what he called the  $P_{odd}$  and  $P_{even}$  functions, which are the partial Patterson functions described above. The interpretation of these functions by Sakurai (1965) for the triclinic modification of quinhydrone ( $C_6H_4(OH)_2, C_6H_4O_2$ ) is close to that above, although more qualitative.

The extension of the structure factor expressions and application of the subcell Patterson function to the 'M=3' case has been discussed by Stephens (1967).

The hkO and hht Patterson projections and the hOt partial (h=0 mod 3 and h≠0 mod 3) Patterson projections were constructed using the room temperature data. These calculations were performed, as were all subsequent Fourier series, using the FORTRAN language computer program MACFOU, written by the author. This program follows several of the suggestions of Rollett (1964) as to a versatile



The Patterson function of selenourea projected on (001).

basic Fourier program. These are the use of the Beevers and Lipson (1934) product form of the series and table look-up of trigonometric functions, variable grid sizes up to 1/120 of the unit cell edge, the possible choice of slant plane projections and sections, and applicability to all symmetries. This last requirement is satisfied by the generation of symmetry related reflections, together with their phase, according to the Waser (1955) formula,

$$F(RH) = e^{-2\pi i \underline{H} \cdot \underline{t}} F(H) \qquad (II-13)$$

for a space group element consisting of a point operation R and a translational component t.

The solution of the hkO Patterson (Figure II-2) is particularly simple in retrospect, although initial attempts to conform to the model suggested by Kondrashev and Andreeva caused difficulties in determining the positions of the nine selenium atoms within the channels. In fact this projection contains nine almost identical regions, each the vector map of an equilateral triangle, with no other major features. This can be interpreted as nine such triangular groups of selenium atoms, equally spaced in this projection. For steric reasons each member of the group of three must be near to c/3 away from the others; this together with the pseudo-symmetry mentioned above yields approximate



Figure II-3. The Patterson function of selenourea projected on ( $\overline{110}$ ). The number of Se-Se vectors contributing to each peak are illustrated. In parentheses are the corresponding values based on the assumed tripled structure.



positions for all selenium atoms. The deviations from this rhombohedral model of the structure were subsequently found to be nowhere greater than one Angstrom, and their nature was a major problem in the structure determination.

The next stage was the solution, within the subcell, of the hh**2** (equivalent to HOL, Figure II-3) and partial hO**2** (h=0 mod 3, HHL, Figure II-4) Patterson projections. Compatible solutions for these were found assuming a near rhombohedral cell containing nine selenium atoms, and the origin for this structure was chosen so that the two-fold axis it contained passed through that origin. The structure then belonged to the space group  $P3_112$  (full cell,  $P3_121$  sub cell), and for some time refinement was carried out on this basis.

Structure factors were calculated for the room temperature data with  $-h+k=0 \mod 3$ . The scattering factors used throughout were taken from International Tables for X-ray Crystallography, and were those of Freeman and Watson (1961) for selenium and Hoerni and Ibers (1951) for carbon and nitrogen. The calculated structure i $\phi_c$ , used to compute the corresponding electron density and difference electron density

$$D(\underline{\mathbf{r}}) = \sum_{\underline{\mathbf{H}}} (|\mathbf{F}_{0}| - |\mathbf{F}_{c}|) e^{i(\mathbf{\phi}_{c} - 2\pi \underline{\mathbf{H}} \cdot \underline{\mathbf{r}})}$$
(II-14)

syntheses. The carbon and nitrogen atoms were assigned positions

based on peaks found on these maps. Least squares refinement of the positional parameters of this model, using the program MACLS ` written by J.S.Stephens, gave an unweighted R factor of about 0.2. The structure deduced at this stage contained spiral chains of selenourea molecules similar to those in the urea and thiourea hydrocarbon adducts.

The displacements giving rise to the true structure were determined in the following manner. The partial hof Patterson  $(h \neq 0 \mod 3$ , Figure II-5) was examined and the prominent pattern of peaks on it were interpreted as arising from chains displaced as units (the within chain vectors are positive peaks) from their average positions, and in opposite directions to their subcell related chains (since the vectors between such chains correspond to minima). One subcell chain type only can be involved, since vectors between different chain types do not fall on features of this map. The vectors involved are illustrated in Figure II-5.

The particular chain involved and the extent of its displacement was found by scrutiny of the xxz electron density projection at this stage. The magnitudes of the electron density for the three independent selenium atoms on this map were 116, 100 and 100 electrons per square Angstrom, where the latter two were related by the assumed 2-fold axis. A further pair of unexplained peaks of magnitude 19  $e/A^2$  were found at a distance of 1.8% from the 100  $e/A^2$  peaks. When one chain was displaced the indicated



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C

distance (the choice was dictated by the indexing of the other projections) the R factor of the hhe projection was immediately reduced from 0.22 to 0.18.

Figure II-6 shows the relationship between the assumed  $P3_1l2$  structure and the true structure. The small cell on the lower left illustrates the symmetry elements of the space group  $P3_12l$ , that on the upper right the one three-fold axis of the three retained in the true structure, which is not the one at the small cell origin. The choice of large cell therefore requires the change of origin illustrated in the figure, to (2/3,1/3) in the small cell co-ordinates. The choice of the symmetry related point (1/3,2/3) as origin leads to the twin structure mentioned above.

#### Refinement.

The model described in the previous section was refined by least squares, using all the room temperature data and an overall temperature factor, to R=0.12. The data was then merged, that is, structure factors common to different layers were averaged, to form a set of only independent structure factors, using the layer scale constants determined by least squares. The individual measurements were weighted in these averages according to their previously assigned standard deviations, and the average values ascribed standard deviations accordingly. Even at this advanced stage of

![](_page_52_Figure_0.jpeg)

Figure II-6.

Relationship in position and symmetry between true unit cell and pseudosymmetric subcell.

refinement, the standard deviations in light atom co-ordinates were still about 0.1Å, and there appeared little hope of gaining worthwhile chemical information.

Some likely reasons for these difficulties have been discussed by Scheringer (1965). Three of the reasons he suggests for ill-conditioned least squares problems are applicable here. Foremost of course, is the smallness of the data set, consisting of 781 reflections, corresponding to projection data only. However, many crystal structures have been solved adequately with only projection data, which suggests other factors are operative. The possibility is always open to us to extend the data set, and this is exactly what we chose to do in the case of the low temperature work.

Unfortunately the other two effects are intrinsic to the selenourea structure analysis. One is the presence of atoms of widely differing scattering power; this is a more serious problem, and in earlier times would have been considered unsurmountable. But yet a third difficulty exists, that is, high correlations between parameters related by the elements of pseudo-symmetry outlined above. In this respect the structure of selenourea is very similar to that of guanidinium aluminum sulphate hexahydrate  $(C(NH_2)_3Ae(SO_4)_2. 6H_2O, commonly called GASH)$  which was chosen by Scheringer as an example of this effect. GASH, a ferroelectric crystal, belongs to the polar trigonal space group P3lm, with cell

constants a = 11.75Å, c = 8.94Å (Geller and Booth, 1959). This cell contains three formula units, and is almost triply primitive, the subcell to supercell transformation being the same as for selenourea. The difficulties encountered in the refinement were discussed in detail by Geller and Katz (1962), and only prolonged least squares reiteration, using only fractional parameter changes, gave a satisfactory structure (Lingafelter, Orioli, Schein and Stewart, 1966). A similar cautious empirical approach has been followed in the case of selenourea.

In the light of these circumstances, it was clearly advantageous to divert attention to the low temperature data. This in turn was refined by least squares, using the room temperature co-ordinates as a starting point, and individual layer scale constants. The final individual layer R factors are listed in Table II-2. The data was then merged as described for the room temperature data, but now reflections occured up to five times, for, in hexagonal axes, such a set as the following are equivalent.

102		hO
102	100	hk2
012		h,h+1,2
012		hle
112	2 6	h,h+2,2

The selenium scattering curve was now corrected for the real part of the anomalous scattering for CuK radiation. Although no attempt was made to discriminate between hkt and hkt reflections

in the data collection, the methods used ensured a similar number of each type, and therefore a systematic error in sclenium z coordinate appreciably less than the possible value of 0.02Å suggested by Cruickshank and McDonald (1957). The final cycles were carried out using anisotropic temperature factors on the selenium atoms, and with the carbon parameters fixed. This was done because high correlations with other parameters caused large shifts to be calculated for several of these atoms, which would have produced chemically unlikely atomic configurations. The detailed carbon positions were chosen on the basis of chemical considerations, although their approximate positions can be derived from electron density maps. This process was continued until a minimum weighted R factor of 0.146 was obtained. The corresponding atomic parameters (with errors derived from least squares) and structure factors are listed in Table II-3 and Table II-4 respectively.

Although the majority of the room temperature light atom co-ordinates were identical, within the errors, to their low temperature equivalents, several had to be adjusted in the light of the low temperature refinement. This was done, the light atom positions were fixed, and a subsequent two further cycles of refinement brought the weighted R factor for this data to 0.116. The final parameters can be found in TableII-5 and the corresponding structure factors in Table II-6.

The final co-ordinates mentioned above were used to compute interatomic distances and angles, using the program ERROR written

# TABLE II-3.

Final atomic parame	ters /	at	±100°	C.
---------------------	--------	----	-------	----

Atom	Number	x	y	<u>Z</u>	B
Se	1 2	0.9318(5) 0.6054(5)	0.9654(5) 0.2893(6)	0.4280 0.5708(6)	
	3	0.2625(6)	0.6397(6)	0.5537(6)	
	. 4	0.2898(6)	0.0325(6)	0.8646(7)	
	5	0.9351(6)	0.3060(6)	0.8564(7)	
	8	0.9417(7)	0.6444(6)	0.1384(8)	
	9	0.6132(6)	0.9727(6)	0.1512(7)	
~	7	0.2622(6)	0.3000(6)	0.1489(7)	
C	2	0.072(6)	0.030(6)	0.426(6)	2.0
	3	0.402(6)	0.671(6)	0.540(6)	2.0
	* 4	0.428(6)	0.056(6)	0.855(6)	2.0
	5	0.076(6)	0.346(6)	0.852(6)	2.0
	6	0.730(6)	0.645(6)	0.851(6)	2.0
	7	0.389(6)	0.414(6)	0.152(6)	2.0
	Ö	0.083(6)	0.727(6)	0.148(6)	2.0
NI	1	0.112(5)	0.079(5)	0.144(0) 0.335(4)	3 3(74)
111	2	0.795(4)	0.425(4)	0.438(4)	1.6(11)
	3	0.454(4)	0.705(4)	0.443(4)	1.9(11)
	4	0.480(3)	0.103(3)	0.769(3)	0.0(7)
	5	0.123(5)	0.363(5)	0.753(4)	3.4(14)
	6	0.782(4)	0.676(4)	0.762(4)	1.8(11)
	2	0.437(3)	0.480(3)	0.075(3)	0.2(8)
	9	0.806(3)	0.089(3)	0.060(3)	0.5(8)
N2	í	0.122(4)	0.041(4)	0.512(3)	0.9(9)
	2	0.810(5)	0.416(5)	0.656(4)	4.1(15)
	3	0.437(3)	0.654(3)	0.631(3)	0.6(8)
	4	0.478(5)	0.063(5)	0.947(4)	4.7(16)
	5	0.126(4)	0.357(4)	0.941(4)	2.0(11)
	0	0.446(4)	0.630(4)	0.923(3)	1.0(11)
	8	0.132(4)	0.720(4)	0.233(4)	2:5(12)
	9	0.799(4)	0.032(4)	0.231(3)	1.9(11)

(Continued overleaf)

### TABLE II-3

## Final atomic parameters at -100°C.

(continued)

Anisotropic temperature factor components (X 104).

Atom	Number	B <sub>11</sub>	B <sub>22</sub>	B_33	B <sub>12</sub>	B <sub>13</sub>	<sup>B</sup> 23
Se	l	14(4)	26(4)	8(4)	13(4)	2(3)	-3(4)
	2	22(5)	14(5)	29(4)	0(5)	-12(4)	-10(4)
	3	17(5)	40(5)	35(4)	14(5)	-6(5)	-17(5)
	4	13(4)	23(4)	29(4)	13(4)	5(4)	9(4)
	5	4(5)	20(5)	29(5)	4(5)	-9(5)	-10(5)
	6 .	11(5)	29(5)	42(4)	11(5)	10(5)	-1(5)
	7	15(4)	40(4)	43(4)	11(4)	-3(4)	-5(4)
	8	18(6)	27(5)	17(6)	6(5)	-3(6)	1(5)
	9	9(5)	20(5)	17(4)	6(5)	1(5)	ı(4)

. 14	L	FØ	FC	PHE	н	L	FP	FC	PHI	H	ι	+0	FC	PHI	н	L	FØ	FC	PHI	HL	Fø	FC	PHE
0	,	506	641	189	****	3456	69 84 92	63 76 72	97 157 281 222	888	6. Te	67	16	142		***	89	42	269		94 63 80	94	276
ceco	12.	98 51 79	86	60 309 309 302	****	7.0.	58 68 67 71	47 67 20 66	191 347 351		10.	90 88 83	40	224		8. 4. 10.	80	34	180		127	122	244
	16	21 30	40	19 352 51	****	12.	73 90 87 82	45	125 26 38 250	2200	1	87 80 80	10	163	444	-7.	81 81	426	301 11 302	567	644	55.884	334
	10	28	27	289 290 308 129	****	16.	77	30	147	3993		101 990 73	31	149 220 218	1444	-3.	88 90 46	667	259 95 64 146	3 10	53 44 75 33	40	192 278 142 30
		41 91 53	33	136		12.	90	21	83 273 271	9999	-5-	69 98 227	60 12 108 249	221		1234	90 122 86 84	100	331 283 331 235	3 13	48 29 25 142	48 36 21 23	104
		215	20	279	75555	-8.	65 60 57	42 56 33	222		-1	409	390	250	1444	7. 1. 8.	80 80	83	102	4-12.	158	299324	123
	2345	182 33 19	12	57 90 142 248	5555		69 39 127	73 38 145	11		3450	244 93 69 /1	265	308 33 92	15	-6.	67 64 82	25	91 42 184 297		184	177	354
	67.89	59 28 38 78	57	228 245 12 312	5555	01-27	37	35	54 262 156 227	9999	7 8.	125	138 48 19 68	2267 267 14	15	-2.	83 81 78 81	2878518	349 336 133 136	4 -3.	225	230 23	228
	10	28	40 40 15	104	2555	4567	107	90 27 26	129 344 308 340	999	12.	80 80	447	58 60 305 287	15	2.3	115 82 84	21 90 53 23	17 319 189 250	••••	66 46 384 138	57 13 464 129	171 146 247 15
	15	463	552232	150	75555	9. 10. 11.	69273	15 43 41	313 288 97	10-10	10	81 85 88	18339	193 99 238	15 16 16	-3.	204	83 38 38	102	5 67.	140	1229	305
~~~~~	-14.	30 32 55	12 40 28 28	39 73 304 262	5556	13.	86 80 78 80	14 8 30	130	10	-1.	13	80	151 225 279 284	16	0.1.	32 65 77 90	31 67	294	• 10• • 11•	153	143	23
~~~~~		25 63 55 20	24 61 49	244 53 63 99	6666	14 130 120	164 83 88 119	134 54 28 108	229 220 187 45	10	-3.	110 69 68 38	114 52 33 34	33 112 152 121	1-	-16.	(= 1 40	32	334	4 13• 4 14• 5-14• 5-13•	151 142 138 148	92	73 155 202 22
~~~~		179	39 69 77 107	130 205 10 271	6000	-9.	166	183	58 118 183 229	10	1234	68 69 84 71	29 85 54	52 341 217 268		15	209 50 48 230	176	174 357 113 349	5-12• 5-11• 5-10• 5-9•	129	35 86 71 33	327 359 186 199
~~~~	-1	19	16 8 28	190	00004	-54	57	204	339 50 258	10	7.0	73	67 33	29		-9-8	107	13	237 37 148 118	5 -7.	109	19 50 17 68	101
~~~~~	3456	151 99 119 80	150 84 110 66	80 141 282	00000	-1	73 134 58	64 128 46	273	10	10.	85 81 78	21 31 43	311	1		48 218 177	58 197 173	279		41 63	42 39 69	282
2222	789	29 46 108 32	32 39 102 29	260 254 246	6666	3456	126	121 211 66 39	276 39 118 103			89 79 82 85	434077	22 343 24 184	1	-2-1-0.	170	192 10 8	139 60 159 301	5 23 4	51 85 56	30	46 145 162 88
~~~~	12	115	120	115	0000	7 9. 10	279 68 71 166	247	233 287 320 71		-8.	88 90 91 73	55	185 260 337	1	2345	113 311 38 104	127 331 33 97	253 352 36 80	5 5 5 6• 5 7• 5 8•	96 103 109 117	85 39 35 37	335 337 176 173
~~~~~	16	23 83 78	20 18 41	45	00004	120	88 83 125	48 125	131 285 245		-1-	643	16259	27 265 267	1	6789	395 37 36 328	40	237	5 10.5 11.5 12.	123	25	163
	13.	88 73 125	45 22 89	196		14.	77 81 86	56	44 302 60		12.3	73	10	94 306 224	1	12	162	32 146 28	273 200 132	5 14• 6-13• 6-12•	138	61 39 23	221
3933	-9.	65 60 235	35 33 263 111	31 160 260 327	1-1-1	10. -9. -8. -7.	73 73 70 67	70	128 247 153 198		5.	73 91 90 88	32	146 96 94		15.	45292	34 47 5 33	94 308 210 200	6-10• 6 -9• 6 -8• 6 -7•	128	62 48 43	118 132 121 258
		235 373	25 211 399 81	326	1111	-60	60	28	205		10.	85 82 79 164	80 245 91	167 288 200	~~~~~	12	104 63 87	101 69 81	18 307 174	6 -6° 6 -5° 6 -3	108 104 98 87	637727	250 78 321 23
	OLV	591 46 76 386	582 33 65	358 18 261	1111	-1	123	125	134 180 340 25	122	-9-	79 81 84	4822	219 201 244	~~~~	-87-65	5527	4725	207	6 -1.	57 28	16	3052
333	4561	276 75 144 334	224 57 104 260	67 187 41 259	1177	3450	65 58 60	73 29 50 43	219 157 238 15	12	-5.	89 91 91	64 18 43	78 89 182 291	~~~~		129	118	334 268 346 153	6 5.	48	45 124 40 80	173 97 337 328
	10	60 65 137 125	50 36 99 84	327 320 82 67	1111	7. 8. 9.	67 70 73	65 43 35 55	159 342 28 335	12	-10	74 76 258 76	47 37 290 46	346 255 121 180	NNNN	01.	123	10	316 269 163 106	6 7. 6 8. 6 9. 6 10.	115	68 61 29 42	307 300 121 260
	13.	116	20 89 44 21	306 253 85	7778-	120	86 81 77	37	155 68 265	12	13456	185	219	308	~~~~	5678	56 50 44	583822	330 253 315	6 12• 6 13• 7-13•	153	41 45 22	218
****	-15-	84 17 82 87	29 24 24 35	28 313 87 303	8-8-	13.	80 83 88	29 37 18 33	75 58 325	12	7 8. 9. 10.	87 84 81 79	42	159 61 196	2222	10.	417	33	294	7-11. 7-10. 7-9.	156	78 26 56 163	187
****	12.	90 84 71 67	9 63 425	34 336 149 241		-9. -8. -7.	73 73 67	67 21 79 58	192 180 226 312	12	10-0-	90 89 82 80	182237	250 227 331	New	13	39329	17 35 48 54	221 161 211	7 -7.	121	41 50 196 182	10
		58	70 80	260	00000	-4-3-21	67 60 102	10	72 31 310	1333	-60	657 897	10466	208	****	32.0	31 44 26	23422	343		312	36 82 338 51	67 58 161
		86 57	92 38 146 39	294		0122	48	29 28	175	1111	-2.0	91 93 93 93 93 93 93 93 93 93 93 93 93 93	329	272 282			38950	3533	95	234	284	337	242
4	1.	120	36	336	8	\$	22 78	68	200	B	3:	31	38	167	3	-30	17	85	23	1 %.	122	223	301

### • UNBBSERVED REFLECTION. FO IS MINIMUM DESERVABLE • ADSORPTION CORRECTION UNRELIABLE

TABLE II-4.

		FØ	FC	PHE	HL	FO	FC	PHI	HL	FØ	FC	PHE	HL	FØ	FC	PHI		FØ	FC	PHI
	Į .	138	173	132	13 =?*	136	159	274	3 15:	21	20	265	3 -1.	58	33	170	3=1?*	54	21	310
	1 11:	129	30	240	13 1.	48 97	34	212	4-15.	24	19	332	9 0.	31	20	231	3-10	250	267	129
	7 13.	139	61	42		94	92	104	-12	42	34	190	9 3.	38	23	221	3 -7	367	386	239
	A-10.	137	21	278	13 %	46	143	298	4 -90	25	28	56	9 6.	40	50	302	3 - 4	232	259	160
	8 -8.	28	11	311	14 -4:	140	38	327	4 -7	48	11	156	9 8.	49	12	209	3 -20	27	11	22
	8 -5.	118	37	108	-1:	105	51	171		110	110	321	10 -4.	40	56	91	3 12	52	50	294
	8 -2.	85	49	223	1: 1:	105	22	38		97	112	130	10 -10	40	18	237	3 4 5	56	62 45 61	192
< •	8 0 A 1. 8 2.	41 69 85	39	142			38 26 50	243	4 2.	16 56	11 58 21	122	10 1	68	57	245	3 67	70	38	299
	8 4.	113	75	215	13 -1:	08 73	32	341	: :	30	39	98	10 5.	40	21	74	3 9.	42	21	319
	A 6.	22	25	341	13 1:	73	51	62	4 7	224	17	314	11 -3.	62	54	319	3 13:	54	40	17
	6 9.	129	50	105	16 -1	104	128	184	11	30	110	212	11 - 1.	104	89	112	3 15.	40	26	4
	8 12.	143	16	264	18 2	82	126	6	: 13:	30	22	113		129	105	290	4-12.	34	22	34
	9-120	140	21	283	1-15+	29	17	220	5-11.	24	34	294	11 *	40 67	26	286	4-10	28	128	17:
	9 -90	158	21	131	-13:	32	30	261	5 -9	13	59	100	12 -3.	76	60	26	4 -7	31	28	200
	9 -60 9 -50	127	80	235	1-11	33	35	47	5 -6	132	117	270	12 0	54	48	195	4 - 3	116	137	346
	9 -3	127	134	23		59	50	180	5 -3	53	40	101 203	12 3.	76	26	21		114	109	176
	9 0	60	15	208		139	108	281	5 0	28	14	179	13 0.	92 45 92	18	322 288	4 23	35 40 38	37 44 46	348
4	9 2.9 3.9 4.	92 76 121	21	268 192 89		46	93 39 43	284	5 3.	151 28	135	102	13 -2.	111 96	24	296	4 40	21 26	39	221
	9 5. 9 6. 7.	124	58	151	ļ ļ.	12	55	240	5 5	294	344 45 28	237	1: 1	41 80	33	342	4 7	52	68	139
	9 8. 9. 9.	129	16	276		162	169	72	5 8	284	337	58	15 -20	75	16	241	\$ 10	45	50	301
÷.	9 11. 9 12.	145	10	119	1 9	109	97 37	255	6-11.	132	164	251 253	15 1.	63	36	117	4 140	29	26	230
	0-10-	146	61	125	1 9	76	66	246	6 -9	48 38	98 51 43	254		(* 3			5-130 5-120 5-110	28 31 32	27	91 235
·	0 -8.	129	61 19 23	331 85 31	1 11	61 97 35	101	148 79 284	6 -7	37	36	354	1-15	38 29 46	24	345	5-10*	33 68 25	27	138
	0 -5.	128	37	177 282 292	1 15	32	21	294	6 -3	32	27	342	1-12	58	53	251	5 -7	63	70	242
	0 -2.	97	37	41	2-14•	186	143	80	6 -1.	29	13	69	1 -9	34	20	297	5 -4	65	69	292
		19	12	295	2-11.	40	19	297	6 23	116	101	172	1 -5	98	94	236	5 -1.	19	21	31
	0 5.	128	37	178	2 -8*	33	11	121	6 5	85	103	328	1 -3	134	123	288	5 2.	168	179	140
	0 7.	129	55	3007	2 -5	355 90 76	363 72 66	181 99 146	6 8.	42 38 40	16	204 40 53		66 33 27	2227	258 315 302	555	120	142	310
		146	46	125	2 -3	214	223 52 19	286	6 10	40 68 49	65 54 25	60 139 268	1 3	108	105	145	5 7.	25	27	33:
	11-10•	138	48	286	2 0	59	22	353	7 -9+	40	27	201	1 5	35	33	239	5 10.	33	42	221
	-7.	156	17	102	234	27	28	202	7 -6"	37	32	296	1 8	67	66	318	5 13.	28	32	15
	1 -3:	129	40	23	2 67	32	39	110	1 -2	33	**	75	1 11.	55	54	139	6 -2.	81	39	79
1	1 -1.	82	65	219	2 10.	102	85	207	1 1	33	27	22	1 14	47	43	232	6 1.	66	49	172
	1 3	100	22	83	2 12	136	131	8 6	1 3	122	138	214	2-150	31	37	239	7 -3	135	125	317
	11 4. 11 5.	129	53	280 300 328	2 140	48 59 27	26 95 23	163	1 6	35	19	314	2-120	36	51	227	7 -1	79	98 19	154
	11 7.	156	18	102	3-14.	31	68	110	1 8.	40	11	333	2 -9	27	14	134	7 2.	102	19	335
-	11 10.	139	12	184	3-11.	29	19	275	8 -9	187	184	111	2 -6	48	12	234	8 -2.	94	44 28	216
1	2 -7.	149	30	352	3 -7.	45	45	38	8 -6	247	236	309	2 -3	101	108	243	8 1.	76 94	38	27
	2 -3	139	56	1402	3 -5	ij	100	206	8 -4• 8 -3 8 -2	163	112	203	2 -1	49 61 58	58	147	8 3. 9 -3 9 -2.	164	180	284
-	2 -2.	101	19	57	3 -1	39 38 31	45	300	8 -1.	35	34	224	2 3	138	132	138	9 -1.	263	276	246
	2 2:	121	35	170	3 0	72	45	153	8 3	48	86	19	200	123	124	33/	9 2.	154	177	135
	3 5	137	37	288		40	38	166	8 5	152	164	371	2 9	13	56	280	10 -2.	101	19	141
	2 7.	49	26	157	3 %	29	34	156	8 8.	137	162	352	212	94	112	123	10 1.	101	24	192
	3 -7.	41	19	126	1 10.	56	51	114	9 -7	40	45	229	2 5:	21	33	287	11 -3:	122	56	11
551	1 -3:	152	20	10	1 1:	36	30	105	3 -4.	19	222	110	5-13:	46	57	237	11 1.	50	42	198
1		FRYE		FLEC	TIAN U	REL	ADLE	IMUM	BSER	ABLE						•				

TABLE II-4 (continued)

	HL	FB	FC	PHI	HL	FE	FC	PHE	HL	FØ	FC	PHI	HL		FC	PHI	HL	FB	FC	PHI
	11 3:	122	64	???	: ?	82	76	157	1-17:	136	52	223	2 2	225	219	237	1 17:	153	34	216
	12 -2.	103	114	368	4 2.	411	495	244	1-10.	120	49	128	\$ 1.	227	276	ili	3-11.	48	59	216
	12 -1.	118	145	270 98	4 5	170	178	263	1 - 7:	117	23	222	\$ 100	44	30	122	2 -9	59	43	236
	13 1:	133	19	154	\$ 7.	41	68	104	1 -3"	103	50	512	3 12.	109	116	249	2 -7	63	33	292
	13 -20	102	33	295	4 9	12	49	190	1 -3.	114	120	235	8-13:	28	18	250	2 -3	78	33	310
	13 0.	13	24	330	- 11:	24	13	238	1-1	119	108	136	6-11	68	62	236	2 -2	56	46	110
	13 -20	102	33	101	4 13.	47	17	316	Į Į.	51	ií	268	6 -9	39	16	215	2 0	54	50	156
	4 -1:	10	20	237	5-14.	27	21	255	1 3.	56	53	44	6 -7.	36	43	225	2 2	148	147	114
	1: 1.	70	109	328	3-11:	32	29	128	1 6.	103	112	303	6 -5.	27	17	21	2 5	102	120	338
			•		5 -90	29	45	295	1 :	117	żş	124	6 -2	112	191	190	2 7	35	57	156
	1=13:	137	32	108	3 -7	65	78	150	1 10:	128	\$3	89	6 0	62	57	346	2 90	40	47	152
	1-12.	158	58	333	3 -4	126	142	335	1 13:	156	51	262	6 3	33	71	182	3 -3	163	99	120
	-90	120	29	164	3 = 7	17	63	226	2-10	144	131	217	6 5	46	50	46	3 -20	66	68	252
	-70	107	54	325	5 9	35	41	52	3 -9.	37	31	269	6 7.	36	39	205	1 1.	66	43	21
	1 -3	218	187	346	5 3	60	62	250	2 -6	126	140	93	6 9.	35	34	113	3 3	172	167	356
	1 -2	130	124	101	3 3	25	75	311	2 -3	29	. 71	294	6 11.	32	32	115	4-12.	28	21	192
	0	98	78	158	3 7	50	12	102	3 -1	66	49	55	1-12.	25	16	277	4 -9	38	35	24
	13	369	137	321	5 10-	32	26	78	2 1	186	176	192	7-1g.	29	29	251	4 -7	33	20	56
	1 3	127	119	86	5 11.	34	23	333	2 3	91	51	128	1 - 1.	58	52	335	4 -5	54	66	37
	1 2:	103	-	231	3 4	27	31	200	2 6	10	56	350	1 -3	64	48	76		37	39	215
	1 10.	146	134	350	6-12.	28	11	262	2 8	236	339	68	1 -3	49	51	283	2 0	59	55	336
	1 12:	128	26	311	6-10	53	56	130	2 11"	102	146	259	1 -1	47	32	136	4 2	23	17	229
	1-130	142	38	167	6 -7	45	37	297	1-13	28	41	16	1 2.	25	14	176	1 1	25	31	214
	2-140	27	38	228	6 -5	58	63	80	3-11	12	52	229	1 5	55	46	72	4 7	57	66	136
	2-12.	32	21	39	6 -3	32	102	300	3 -9	43	60	289	1 1	63	76	31	4 10	33	33	298
	2 -90	25	11	240	6 0	57	37	224	3 -6	23	11	241	1	31	63	297	2 12.	28	48	233
	2 -1	58	46	138	6 2 3	26	13	357	3 -3	85	85	334	1 11:	27	30	152	5-13.	28	12	288
	2 -4	151	163	343	6 5	25	32	352	3 -1.	12	82	153	8 -3	243	238	153	5-11	46	50	227
	2 -2	77	82	33	6 7	43		255	1 1	63	04 147	312	8 0	40	31	24	5 -80	36	15	292
	2 0	24	29	311	6 10	33	14	108	3 3	56	62	244	8 2.	100	68	173	5 -6	58	74	47
	2 2	32	80	160	6 12.	28	90 13	132	3 6	85	100	336	9 -3.	124	26	196	5 -3.	27	17	320
	27	86	86	313	7 -30	60	34	298	1 8:	25	24	53	9 0.	50	16	100	3 -1	29	25	287
	2 7	27	25	35	7 -1	194	227	63	3 10.	33	22	181	9 2.	124	16	273	5 2	36	24	177
	2 10-	26	11	164	1 1.	195	210	323	3 13.	28	22	279	$10 - 3 \cdot 10 - 2 \cdot 1$	119	35	299	5 3	27	20	29
	3 13.	32	13	338	8 -3.	63	13	354	4-14.	27	12	158	10 0	49	40	318	5 6.	29	1	59
	2 15.	40	48	217	8 -1.	80	39	210	-11:	32	38	206	10 3.	119	36	235	5 8	36	30	113
	3-13.	32	33	48	8 2.	97	17	37	4-10.	36	15	228	11 -1.	121	37	123	\$ 10	34	34	217
	3-11.	36	99	254	9 -3	82	104	27	4 -7	52	47	161	11 1.	162	145	300	5 13	36	29	31
	3 -9	20	34	117	9 -1.	82	32	205	\$ -3	109	116	327	12 -2.	45 79	39	225	6-11	103	87	280
	1 -9	34	24	33	2 2.	100	11	311		.63	62	192	13 1.	19	27	108	6 -8.	110	39	35
·	3 - 3.	20	60	324	10 -2.	123	8	94	4 9	60	56	100	13 -1:	15	22	51	6 -6.	44	36	351
	3 -1	92	32	115		50	21	155	1 3	63	50	138	13 1:	31	40	333	6 -4	158	172	314
	1 1	31	38	159	10 2.	123	IÓI	332	4.5	13	94	327		(= )	24	124	6 -1	123	77	246
	3 3	33	18	64	11 -2:	110	39	209	4 7	19	32	353	1-13.	143	01	359	6 1.	43	32	135
21	3 6	51	62	288	11 0	50	46	50	4 10	45	30 62	200	1-11.	158	34	140	6 4	167	177	301
	3 8	14	\$7	118	11 3:	118	31	341	: 13:	32	ži	172	1 -90	121	56	352	6 6	197	38	234
		131	133	180	13 -1:	107	52	352	2-13	169	145	183	1 -7.	115	19	124	6 8.	52	20	172
	1 15.	32	31	357	13 1:	88	25	154	3-11.	31	ij	315		98	15	140	6 11	81	63	81
	4-11:		H	217	13 - ?*	134	45	315	5 -9	44	48	189	1 -1.	70	47	138	1-11	61	61	230
* .*	3-11.	34	16	127	1 1.	76	.19	130	3 -0	137	125	354	1 1.	57	31	122	7 -9.	48	37	38
	4 -90	44	19	326	14 -1:	66	24	210	3 - 1.	40	12	189	12.	46	28	322	1 -6	56	55	264
	4 -6.	61	69	11	1: 1:	37	39	10	3 -1.	204	211	122	1 5.	104	37	151	1 -3"	29	10	278
	4 -4	269	585	227	14 24	66 K- 1	40	312	3 1	227	217	309	1	131	33	11	1 -1	43	38	176
	9 -2 9 -1	32	\$39	203	1-13-	148	14	81	3 5.	39	30	122	1 11:	130	37	272	11	15	41	232
		SERY	ED R	FLEG	ILen.	F		EMUM	BASER	ABLE										

TABLE II-4 (continued)

H L FO FC PHI H L FO	FC PHI H L F	FC PHI	HLI	FØ FC	PHI H	L FØ	FC	PHI
7         3         61         45         258         5         -4         57           7         4         107         112         23         5         -2         56           7         4         107         112         23         5         -2         56           7         4         40         157         5         -2         51           7         6         36         18         146         5         -1         26           7         7         51         57         159         5         1         36           7         9         33         17         67         5         3         63           7         10         30         40         236         5         3         63           7         11         30         40         236         5         3         63           8         -0         43         12         141         5         5         3           8         -0         43         12         141         5         5         3	53         2         1         10         6           19         67         2         8         0           73         118         1         2         8           73         25         17         5         10           73         118         1         4         11           75         77         1         5         11           75         205         6         12         16           16         37         1         6         12           64         300         1         8         12           55         234         100         18         12           55         174         110         15         50	9 27 326 5 38 67 5 109 310 8 40 298 2 60 129 6 54 272 9 35 114 7 37 92 1 12 123	9 75 10 -5 10 -6 10 -2 10 -1 10 0 10 1 10 2 10 3	47 42 67 52 67 52 77 52 20 69 64 51 65 67 77 65 77	173 9 145 9 87 9 316 10 - 261 10 273 10 7 10 163 96	3 95 104 50 57 104 57 63 10 26 10 26 10 26 10 54 10 54	92 653 5051	188 296 355 36 37 219 256
BB       30       35       3400       5       9       466         B6       347       39       100       5       9       466         R5       341       5       10       5       9       466         R5       39       103       5       10       275         B6       79       113       5       10       275         B6       79       73       113       6       -10       36         R5       337       317       36       5       11.2       275         R1       87       79       365       5       11.2       275         R - 1       87       347       50       6       -14.0       36         8 - 0       271       6       -14.0       36       6       -14.0       37         8 - 0       327       205       6       -14.0       37       37       37       37       37       37       37       37       37       37       37       37       37       37       37       37       37       37       37       37       37       37       37       37       <	640       12       14       12         640       12      8       4       14         640       12      6       14       14         641       12      6       14       14         642       37       14       34       14       14         728       932108       14       33       14       34         728       932108       22       -4       14       34       109         728       932108       22       22       14       33       14       33       14       33       14       33       14       34       109       14       14       14       109       14       14       109       14       14       109       14       109       14       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       109       1	3142014 42342014 42342014 42342014 42342014 42342014 42342014 42342014 423214 42342014 423214 142342014 1123451 234512000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 1122144122000 11221441220000 11221441220000 11221441220000 11221441220000 11221441220000000000	LO 55 K- L-2-L 10- L-10- L-10- L-1-5- L-1-5- L-1-5- L-1-5- L-1-5- L-1-5- L-1-5- L-1-5- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1- L-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-1-5- L-1-	60 48 61 65 2 8 7 7 5 5 5 0 2 2 7 7 5 5 5 2 8 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5	26 57 11 12 12 12 12 12 12 12 12 12	10000000000000000000000000000000000000	2154 2163172155600965644576158776653	20180 221508741434811740800940933524430681244306879690 131443088124430681332016879690 131443088124430681332016879690
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{c} 2 \\ 2 \\ 2 \\ 7 \\ 7 \\ 2 \\ 7 \\ 7 \\ 2 \\ 7 \\ 7$		377 29 341 507 341 507 3415		3° 6° 2° 101 1° 83 1° 83 2° 101 1° 83 2° 104 3° 78 1° 83 2° 123 1° 85 1° 1° 85 1° 1° 85 1° 1° 1° 1° 1° 1° 1° 1° 1° 1° 1° 1° 1° 1	3 24554 44418025105505965 14441473262333215	1 121 233 245600028525709 2033 245600028525709 233 21232 223 23
$ \begin{bmatrix} 3 & 120 & 121 & 260 & 0 & 0 & 35 \\ 1 & 5 & 134 & 105 & 9 & 286 & 0 & 1 & 37 \\ 1 & 5 & 134 & 134 & 103 & 8 & 2 & 36 \\ 1 & 7 & 121 & 31 & 175 & 6 & 4 & 35 \\ 1 & 7 & 121 & 31 & 175 & 6 & 4 & 35 \\ 1 & 9 & 148 & 174 & 241 & 8 & 6 & 54 \\ 1 & 9 & 148 & 174 & 241 & 8 & 6 & 54 \\ 1 & 9 & 148 & 174 & 241 & 8 & 6 & 54 \\ 1 & 9 & 148 & 174 & 241 & 8 & 6 & 54 \\ 1 & 9 & 148 & 174 & 241 & 8 & 6 & 54 \\ 1 & 9 & 148 & 174 & 241 & 8 & 6 & 54 \\ 1 & 10 & 129 & 59 & 455 & 87 & 400 \\ 1 & 120 & 149 & 53 & 99 & 89 & 49 & 49 \\ 1 & 130 & 139 & 56 & 221 & 8 & 104 & 71 \\ 2 & -100 & 49 & 32 & 156 & 9 & -8 & 68 \\ 2 & -7 & 77 & 71 & 178 & 9 & -5 & 68 \\ 2 & -7 & 77 & 71 & 178 & 9 & -5 & 68 \\ 2 & -7 & 35 & 160 & 53 & 9 & -3 & 61 \\ 2 & -3 & 61 & 53 & 59 & -1 & 53 \\ 2 & -1 & 70 & 59 & 172 & 9 & -1 & 53 \\ 2 & -1 & 70 & 59 & 172 & 9 & 1 & 106 \\ 2 & -7 & 61 & 43 & 240 & 9 & 3 & 71 \\ 2 & 2 & 77 & 65 & 166 & 9 & 4 & 109 \\ \end{bmatrix} $	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	867942245847506654325706 21221212121212121212121212121212121212		79 127 202 243 79 127 244 2050 244 243 760 244 380 99 633 55 99 63 80 77 77 80 77 77 77 77 77 77 77 77 77 77 77 77 77	23797877778888888889999 237978777788888888889999 2379787777888888888899999 2379788888888888999999999999999999999999	1 • 10 + 8 + 8 + 7 + 7 + 10 + 8 + 4 + 7 + 2 + 10 + 8 + 4 + 7 + 2 + 12 + 12 + 12 + 12 + 12 + 12 +	6378834484601409130012630 12314545231633012630	1913 300488372205729201100451 302487572792020100451 302910100451 19557279202000451 195572792000451 195572792000451 195572792000451 195572792000451 195572792000451 195572792000451 195572792000451 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 1955727920004 19557279200000000000000000000000000000000
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	777776-65 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 865 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 855 8-	626 375 837 837 837 837 8437 837 8437 837 837 837 837 837 837 837 837 837 8	2714           370           1-1-           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372           1372     <	K= 1 0. 138 9. 146 8. 151 6. 159 5. 130 4. 129 3. 64 2. 100 1. 82 0. 63 1. 82 0. 63 1. 82 0. 63 1. 59 7. 156 8. 159 7. 159 6. 159 7. 159 6. 159 7. 159 6. 159 7. 159	4421359819990703305739777 4421359819990703305739777	2 3640718464640439667321212121212121212121212121212121212121

. UNDOSERVED REFLECTION. FO IS MINIMUM DESERVABLE B ABSORPTION CORRECTION UNRELIABLE

TABLE II-4 (continued)

H	L	FA	FC	PHI	н	L	FØ
~~~~~	2** 5*** ******************************	446912101284676767641129428097127846767677841129428097120507778	33494535143552554523212424 454133	924339445536399644553639964556399645563996445556399644555363996445553639964455556595622012805399652201328053996522013280539965220132805399652201328053996522013280539965220132805399652201328053996522013280539965220132805399652201328053996522013280539965220132805399652201328053996522013280539965220132805399652201328053996522001328052200132805220005328052200053280522000532805220005328052200053280522000532805220005328052200053280522000532805220005328052200053280522000532805220005328052200053280522000532805220005328052200053280522000532805220005328052200053280522000532805220005328052200053280522000532805220005328052200053280000000000			- 444087375740601064050467876657876 5
	-9. -8. -7. -5. -5. -3	(= 14 141 144 153 157 157 159	8 27 17 78 29 20 103	148 147 141 358 243 166	11122222	0• 1• -2• -1• 0• 1• 2•	36 73 88 75 63 63 63 63
		12075173944479565767495552784875999507479	3353245368562262713887362341420579962388 11111240344616344822848	313627 9934178997789977899778997789977899778997789		-2• -0• 10• 2•	
 		14457477387504750 11457477387504750 115494961112571225759134965534135 110848071257534965534135	97655938896443516655847785L3198174388 10121546545847785L3198174388 10121546545847785L318474388	22211929008147733246458862795959214881554109279592148845886279592148845886279592139257289			

• UNØBSERVED REFLECTIØN. FØ IS MINIMUM ØBSERVABLE 8 ARSØRPTIØN CKRRLETIØN UNRFLIABLE

### TABLE II-5.

Final atomic parameters at 20°C.

Atom	Number	x	У	z	Atom N	lumber	x	у	Z
Se	1	0.933(2)	0.968(2)	0.428	NL	l	0.11	0.08	0.33
	2	0.604(2)	0.292(2)	0.569(2)		2	0.80	0.43	0.49
	3	0.264(2)	0.638(2)	0.559(2)	1 84	3	0.45	0.71	0.45
1986.00	4	0.283(2)	0.974(2)	0.862(2)	1	4	0.48	0.10	0.77
	5	0.941(2)	0.307(2)	0.857(2)	- Car	5	0.12	0.36	0.76
	6	0.606(2)	0.636(2)	0.857(2)	4 8.3	6	0.78	0.68	0.76
- inte	7	0.269(2)	0.306(2)	0.148(2)		7	0.43	0.48	0.07
	8	0.940(2)	0.643(2)	0.140(2)		8	0.13	0.79	0.08
	9	0.610(2)	0.973(2)	0.146(2)		9	0.80	0.08	0.06
C	l	0.07	0.04	0.43	N2	l	0.12	0.04	0.51
- 2	2	0.74	0.39	0.57		2	0.80	0.41	0.66
1	3	0.40	0.67	0.55		3	0.44	0.65	0.63
	4	0.43	0.06	0.86	Sec.	4	0.48	0.06	0.94
	5	0.08	0.35	0.85		5	0.13	0.35	0.94
1	6	0.73	0.65	0.85		6	0.78	0.63	0.92
-	7	0.39	0.41	0.15		7	0.44	0.44	0.24
	8	0.08	0.73	0.15	Sec. 1	8	0.13	0.72	0.24
	9	0.75	0.04	0.15		9	0.80	0.03	0.23

Overall temperature factor = 4.6(1)

.

	310	TISH NAVEFIARTE	· MARRENALIAN CONNEC
926         91         61         00-5           661         26         61         01-5           926         19         01-5         91           927         19         01-5         91           928         91         91-5         91           926         10         91         91-5           927         91         91         91-5           928         12         65         21-5           927         15         45         21-5           927         15         45         20           928         15         45         0           927         02         02         02         02           928         02         02         02         02         02           928         02         02         02         02         02         02           928         02         02         02         02         02         02         02           928         02         02         02         02         02         02         02           946         22         02         02         02         02         02 <th>901         90         60         4         2         601         52         4           2201         12         90         90         2         401         12         4           2201         12         90         90         2         62         6         12         4           2202         612         101         9         2         162         611         4           922         621         11         1         2         2         6         1         4           922         621         41         2         2         6         9         4         4         7         14         4         7         6         1         4         7         14         4         7         6         1         4         4         14         14         4         14</th> <th><math display="block">\begin{array}{c ccccccccccccccccccccccccccccccccccc</math></th> <th>426         46         94         1         9           661         62         66         1         9           911         64         92         1         9           911         64         1         6         1         9           911         64         1         4         1         9         9           921         84         29         6         9</th>	901         90         60         4         2         601         52         4           2201         12         90         90         2         401         12         4           2201         12         90         90         2         62         6         12         4           2202         612         101         9         2         162         611         4           922         621         11         1         2         2         6         1         4           922         621         41         2         2         6         9         4         4         7         14         4         7         6         1         4         7         14         4         7         6         1         4         4         14         14         4         14	$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	426         46         94         1         9           661         62         66         1         9           911         64         92         1         9           911         64         1         6         1         9           911         64         1         4         1         9         9           921         84         29         6         9
1922       61       1222       00       64         1222       60       0       64       1222       1222         1222       60       0       64       1222       1222       1222         1222       60       1222       1222       1222       122       1222       1222         1222       122       122       122       122       122       122       122         1222       122       123       121       121       121       141       141         1012       122       122       122       122       123       123       123         1022       122       123       124       141       141       141       141         1022       122       123       124       143       144       144         1022       122       122       124       144       144       144         1022       122       122       124       144       144       144         1022       122       124       124       144       144       144         1022       122       124       124       144       144       144       144	0001       250       656       97       256       657       97         0001       250       656       97       256       257       256       97         001       657       97       257       257       97       97       97       97         001       57       97	$ \begin{array}{c c c c c c c c c c c c c c c c c c c $	YEE       26       y1       *11-*         402       y1       *21-*         602       6       y1       *21-*         602       6       y1       *21-*         603       6       y1       *21-*         604       6       y1       *21-*         605       6       1       *21-*         607       6       1       *21-*         608       21       *       *         609       9       1       *         601       9       9       *       *         601       9       9       *       *         602       91       1       *       1       *         603       01       92       42       01       *       *         604       92       92       42       01       *       *       *       *         604       92       42       01       1       *       <
162       14       05       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8-       4-       8- <t< th=""><th><math display="block"> \begin{array}{cccccccccccccccccccccccccccccccccccc</math></th><th><math display="block"> \begin{array}{cccccccccccccccccccccccccccccccccccc</math></th><th><math display="block">\begin{array}{cccccccccccccccccccccccccccccccccccc</math></th></t<>	$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$
2       2       4       6       4       6       7	1	C     C <th></th>	
12       0       12       0       12         691       12       0       12       0       12         691       12       0       12       0       12         140       52       0       0       12       0       0         141       52       0       0       12       0       0       12       0       0       12       0       0       12       0       0       12       0       0       12       0       0       12       0       0       12       0       0       12       0       12       0       12       0       12       0       12       0       12       0       12	102       0       1	0       1       1       1       1       0       1       0       1       0       1       0       1       0       1       0       1       1       1	1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1       1     1     1     1     1     1

TABLE II-6

15

.

HL	FØ	FC	PHI	H	L	FØ	FC	PHI	
2 -9	106	102	,11		6.	10	29	152	
3 -40	19	ii.	171	8	8	33	20	265	
3 -1.	148	140	115	10	0.	ŽÖ	12	271	
3 1	142	150	305						
2 2	19	21	133	2	0.	22	191	196	
3 2	123	198	320	- 1	8.	29	29	324	
3 6	170	164	133	7	0.	22	13	270	
\$ 10-	19	15	341	99	-4.	11	20	262	
3 3.	30	13	188	99	-2.	18	46	268	
7 0.	28	21	156	290	-1-	30	ł	100	
10 0.	22	9	263	99	3	34	31	184	
12 8:	22	15	111	10	0.	20	1	46	
1. 0.	20	10	264			- 10		107	
1 0.	21	12	69	23	0.	25	12	228	
3 8	161	170	237	\$	0.	22	20	238	
5 0	28	21	223	?	0.	21	12	301	
6-11	45	32	297	9	ŏ.	žo	ž	26	
6 -90	19	13	119		*	. 11	,		
6 -7	86	81	235	2	0.	33	20	124	
6 -4	102	100	54	4	0.	22	16	134	
6 -2	34	34	249	4	0.	21	ió	253	
6 1.	84	17	194			- 12			
6 3	.13	19	20	1	0.	22	13	338	
6 5.	12	"JI	358	3	0.	66	63	217	
6 7	117	117	115	5	0*	21	26	43	
6 10	19	57	303			- 13			
§ 12.	17	26	110	1	8.	22	12	.59	
8 0	22	13	250	3	0.	Ži	21	82	
10 0:	22	15	250	5	Ŏ.	ŽÔ	ĪŎ	244	
12 0	33	35	19		0.0	= 14		115	
1 0	39	38	154	ŝ	0.	21	17	70	
3 0	22	13	214	4	0.	19	3	149	
5 00	30	8	49			20		145	
7-10.	17	19	183	ż	ŏ.	ŽŎ	š	39	
1 -7.	86	89	299			- 16			
1 -5	96	103	125	•	0.	14	12	•3	
7 -30	19		20						
7 -1.	19	23	37					1	
1 2	19	13	242						
1 5	19	15	186						
7 9.	19 28	1	95						
7 9	18	8	249						
8 0.	22	iş	86						
 10 8:	21	11	282				1.1	1 .	
	K= 8								
1 0.	22	H	141						
3 8	28	26	311						
6 0	29	30	313						
8 -8.	17	3	41						
8 -6	23	20	179						
8 -40	19	16	357		`				
-1	19	16	192						
8 2.	82	94	300						
8 3.	19	10	234						

UNDBSERVED REFLECTION. FO IS MINIMUM OBSERVABLE

TABLE II-6 (continued)

by J. S. Stephens. The effects of correlations were ignored, save for those resulting from the trigonal symmetry, and to the choice' of origin (the z co-ordinate of Se 1 was fixed throughout both refinements at 0.428, which centred the layer at about 0.5). The resulting bond lengths and angles will be presented and considered in detail in the Discussion.

#### Ferroelectricity.

Although the crystal structure found for selenourea is totally unrelated to that of thiourea, we chose to examine selenourea for the presence of ferroelectricity. A crystal belonging to the point group  $C_3$  must necessarily have its dipole moment parallel to <u>c</u>. If the structure of such a crystal is close to an equivalent structure with this dipole moment reversed, so that small atomic displacements might take one into the other, we have the possibility of ferroelectricity, the ferroelectric axis being <u>c</u>. This situation exists in the selenourea structure, if we choose to call displacements of 1.8% small. The relationship between these equivalent structures is exactly that between the twins already remarked as existing in large selenourea crystals; their nature will be described more fully in the Discussion.

Since we have determined the atomic arrangement within the selenourea crystal, we are in a position to estimate the spontaneous polarisation, P<sub>s</sub>, for selenourea. This estimate, however, cannot

be highly precise, because the contribution of any one molecule to the total dipole moment is very small (and is not always of the same sign), the Se-C axis being always nearly perpendicular to <u>c</u>. If we use the low temperature atomic co-ordinates and assume a dipole moment of about 5 Debyes for selenourea (the dipole moments of urea and thiourea are 4.56D and 4.89D respectively, Kumler and Fohlen, 1942), we obtain a value for P<sub>s</sub> of 1300 e.s.u. (4 X 10<sup>-7</sup> coulombs/cm<sup>2</sup>), with +<u>c</u> as the negative pole. Although this is an order of magnitude less than for thiourea it still represents a readily measurable polarisation. The observed pyroelectric and piezoelectric effects (Kondrashev and Andreeva, 1963) and the tendency of the crystals to collect and retain surface dust lend credence to the suggestion that the dipole moment is not vanishingly small, which it could well be within the errors of the structure determination.

To examine the possibility of ferroelectricity, several large crystals of selenourea were ground so as to produce smooth faces, perpendicular to <u>c</u> and about 0.3 cm. apart. These crystals were of uniform cross-section of about 0.1 cm<sup>2</sup>. Silver paste electrodes were applied to the ground surfaces.

The apparatus used for the ferroelectricity measurements follows the design of Diamant, Drenck and Pepinsky (1957). The proper operation of this bridge was checked with samples of triglycine sulphate, a known ferroelectric, prepared in the same way as above. The behaviour of selenourea was then examined from room temperature to  $-130^{\circ}$ C, by the simple expedient of suspending the sample within the low temperature Weissenberg camera. No sign of ferroelectricity was detected at any temperature within this range.

#### CHAPTER III.

#### DISCUSSION.

#### Space group and lattice parameters.

I shall begin by remarking upon one curious feature of selenourea, namely the point group,  $C_3$ . According to Buerger (1956) no organic representative of this point group is known. Although a few examples do in fact exist ( $\propto$ -L(+)-menthol, Ramsay and Rogers, 1952;  $\delta$ -glycine, Iitaka, 1958), this crystal symmetry is certainly a rare one.

The lattice parameters found for selenourea are here compared to the previously reported values.

a(A)	<u>c(Å)</u>	c/a	Reference.
15.285(5)	13.007(5)	0.8510(4)	20°C
15.201(5)	12.950(5)	0.8517(4)	-100°C
15.37(7)	13.08(5)	0.851	Kondrashev & Andreeva
15.34(2)	12.99(2)	0.847	Perez Rodriguez et al

The results of Perez Rodriguez et al., which they claim

to be more precise, are probably more accurate than those of Kondrashev and Andreeva, since the former calibrated their films with the diffraction pattern of a copper wire, while the latter make no mention of a film shrinkage correction, and assign errors of a magnitude to be expected for this effect. Indeed the present work is in excellent agreement with Perez Rodriguez et al. as far as the <u>c</u> axis length is concerned. The results suggest that their <u>a</u> axis length may be slightly in error, in view of the agreement of our value of the axial ratio (which is independent of shrinkage correction) with the Russian work. Measurement of this ratio on an uncalibrated film (that used for the low temperature lattice parameters) gave the result c/a =  $0.850_3$ , which agrees with the previous value within the errors of the determinations.

The discussion of the thermal expansion coefficients will be reserved until after the description of the structure.

#### Molecular structure of selenourea.

The fractional co-ordinates listed in Tables II-5 and II-6 do, in fact, correspond to nine distinct selenourea (V) molecules per asymmetric unit. The bond lengths and angles for each such molecule, derived from the low temperature refinement, are shown in Table III-1.

However, in general, these would not be the best estimates

### TABLE III-1

Dimensions of the individual selenourea molecules.

	Bond Len	gths (Ang	stroms)	Bond Lengths (degrees)			
Molecule	Se-C	C-N1	C-N2	Se-C-Nl	Se-C-N2	N1-C-N2	
1	1.85(8)-	1.33(4)	1.34(7)	115(6)	118(6)	127(6)	
2	1.86(7)	1.36(4)	1.36(7)	120(7)	131(8)	108(6)	
3	1.94(9)	1.44(4)	1.37(6)	120(5)	111(5)	129(6)	
4	1.83(8)	1.35(6)	1.39(4)	126(5)	113(6)	121(6)	
5	1.91(7)	1.43(4)	1.38(4)	117(7)	119(6)	124(5)	
6	1.83(7)	1.34(5)	1.28(6)	119(6)	128(7)	113(5)	
7	1.84(11)	1.34(7)	1.35(7)	128(5)	121(6)	111(7)	
-8	1.87(11)	1.38(7)	1.36(7)	119(4)	119(5)	122(7)	
9	1.83(11)	1.36(7)	1.35(7)	123(4)	115(5)	121(7)	
Average	erage 1.86 1.36		36 `	120	120		
we could make of the molecular dimensions, since it is well known (see, for example, Grenville-Wells (1965)) that the main thermal ' motion of molecular compounds is libration about the centre of mass. Thermal motion has the effect of shortening the apparent bond length, but this can be corrected for (Busing and Levy, 1964) if the details of the motion are known. It is common practice in the particular case of rigid molecules (molecules without rotating groups) to ascribe as much of the motion as possible to translation and libration and correct for the latter (Cruickshank, 1956). In view of the reduction of this effect by use of low temperatures, and the high standard deviations both in the bond lengths themselves, and in the thermal parameters, this correction was not carried out, being insignificant and inaccurate.

The question now arises as to just how accurate an estimate of the molecular dimensions of selenourea can be made on the basis of this study. The improvement in resolution at low temperatures, discussed in the Introduction, is well illustrated by Figure III-1. Although this use of low temperatures and the three-dimensional data favour an accurate determination, the inherent inaccuracies of the intensity measuring technique, the problems in accurate refinement posed by pseudosymmetry and differences in scattering power as discussed in the Experimental, and the additional difficulties of lack of a centre of symmetry and the large number of parameters required to describe the structure, lead to standard

Figure III-1



<u>A</u>.



B.

A portion of the electron density for selenourea projected on (010). A. At -100°C. B. At 20°C. Contours at 5e/A, zero contour broken.

deviations in individual bond lengths which compare unfavourably with the more careful of current work. On the other hand we have in the crystal structure of selenourea a unique situation, certainly as far as molecular crystals are concerned, namely the presence of nine crystallographically distinct molecules in the unit cell.

Kitaigorodskii (1961) notes that organic crystals containing crystallographically distinct molecules, although rare, are not unknown. However the majority of these contain only two distinct molecules, and, as far as I am aware, the previous record was held by trithiocarbonic acid (VI), (Krebs and Gattow, 1965) which contains 4 molecules in the asymmetric unit, 16 in the unit cell of space group  $P2_1/a$ .\* The structure was determined by use of low temperature (-100°C) projection data, and did in fact show that



\* 1:3:5-benzenetricarboxylic acid has 6 molecules per asymmetric unit, 48 per unit cell. D.J.Duchamp and R.E.Marsh (1965). Paper E8, A.C.A. Meeting, Gatlinburg, Tennessee.

the four molecules were essentially identical in structure. However Krebs and Gattow drew the dubious conclusion that the C-S' single bonds were non-equivalent. This, and other studies of crystallographically distinct molecules, such as the low temperature structure of thiourea (VII) (Goldsmith and White, 1959) and the metastable orthorhombic modification of acetamide (VIII) (Hamilton, 1965), confirm the idea that the crystalline distortion of rigid molecules must be extremely small. Thus we can assume that an average molecular structure, based on all nine molecules, is a viable concept.

The structure, bond lengths and angles of this average molecule are displayed in Figure III-2. These are most conveniently compared with the low temperature structures of the analogues urea and thiourea, since in each case librational shortening of bond lengths is reduced. Goldsmith and White's values have been averaged over the two molecules. It is clear that the main difference is simply the increasing length of the C = X bond, and that the remainder of the molecule remains virtually identical throughout the series.

If we can now give support to the C = Se bond length quoted in Figure III-2, we can perhaps justify the choice of carbon positions used in the low temperature refinement. Figure III-3 shows suggested bond length bond order relationships for C-X bonds. If the single and double bond lengths are fitted by the Pauling





Suggested bond order - bond length relationships for carbon - chalcogen bonds. (Semi-logarithmic plot).

X References of Figure III-2.

• Average values listed in Sutton (1959). ▲ Sum of covalent radii given in Pauling (1960). (1960) expression

$$D(n) = D(1)-0.71 \log n$$
 (III-1)

good agreement is found for CXY compounds assuming bond orders of 2.4. These graphs suggest C-X bond orders of about 1.5 throughout the urea series. The actual value is not so important as the support given to this determination by the identity of molecular structure found within the series. Unfortunately there seems to be no theoretical predictions of bond lengths in selenourea, and so we can only compare the experimental values to other members of the urea series, as done above, and to substituted selenoureas (Table I-2) with which the agreement is satisfactory.

#### Crystal structure of selenourea.

The unit cell of selenourea contains nine spiral chains. Each chain is essentially similar in form, since it contains three molecules and runs parallel to the c axis. A typical chain is shown in Figure III-4, which also illustrates the hydrogen bonding within the chain. Each molecule has one hydrogen atom directed towards the molecule above and similarly one towards the molecule below. In this respect they are similar to the spiral chains in the urea- and thiourea- hydrocarbon adducts discussed in the

## Figure III-4

A typical chain in the selenourea structure.

С



Introduction. The relationship between these structures will be discussed in a subsequent section.

The distribution of the molecules within the chains is described in Table III-2. The symbols A. B. C and D refer to the detailed relative z co-ordinate of the particular chain; they fall naturally into these four groups each with a fairly constant separation, in alphabetical order, of 0.6%. That is, the difference between A and D which distinguishes one subcell from the other two is about 1.8%. These displacements are required to remove selenium atoms on neighbouring chains to their van der Waals contact distance of about 3.8%. These contacts can be seen in Figure III-5, which also clearly shows the nine distinct spiral chains. The selenium atoms in contact form an approximately equilateral triangle within the layer, as do, for example, Sel', Se4 and Se7''. Three chains (those labelled A, D, and D, in Table III-2) spiral around the three distinct screw axes in the unit cell. The remaining six chains form two crystallographically equivalent sets, labelled B and C. Each of these chains contains three molecules unrelated by . symmetry, yet each chain comes very close to containing the symmetry element 3, as can be seen in Figure III-5. The great similarity of these chains of unrelated molecules is even better illustrated by Figure III-6, which shows the structure projected down [110], in which case these chains come close to being superimposed. The A, B, C, D relationship can also be clearly seen in

## TABLE III-2.

Distribution of molecules within chains in the selenourea structure.

Chain	Molecules
A	Ml
	MI Ml
В	м6'
	M5''
	M4
C	M7
	M8 '
	M9''
D	M2''
т	M2
	M2 *
Da	M3"
2	M3
	M3'

The positions related by the elements of the space group  $P3_1$  are denoted throughout as

Figure III-5.



The crystal structure of selenourea projected on (001).



this projection. As far as chains B and C are concerned the differences in z co-ordinate within the chain are only just significant. Distortions of similar magnitude seem to exist perpendicular to the chain direction, but these are difficult to estimate quantitively, since the chain axes themselves are displaced from their ideal positions. This is the result of slight relaxations of the structure to permit more efficient packing. The chain  $D_1$  and its environment are rotated slightly counter-clockwise, while the chain  $D_2$  and the molecules surrounding it are rotated clockwise. This results in a slight lateral displacement of chains B and C.

Since each selenourea molecule has a remarkably similar orientation with respect to its chain axis, the existence of the hydrogen bonds within the chains is quite clear cut, their average length being 3.51Å.

The nature of hydrogen bonding between the chains is much more equivocal. We require the N...Se vector to be reasonable both in length and direction. The shortest distance of this type is N27''...Se5(3.46Å) which is similar in length to the intrachain hydrogen bonds. Distances up to 3.75Å (the longest intrachain bond length is 3.68Å) are listed in Table III-3B. These have been examined with respect to the second criterion, and the Table indicates which are proposed as hydrogen bonds. (The others are mainly between molecules in different layers as well as different chains). It can certainly be said that not all nitrogen atoms are involved in interchain

# TABLE III-3

Intermolecular distances in the crystal structure of selenourea.

A. Intrachain hydrogen bonds.		B. Other SeN distances		C. SeSe distances.		C. SeSe distances.	and states		
Se	N	Length (A)	Se	N	Length (Å)	Se	Se	Length at -100°C (Å)	Length at 20°C(Å)
1 2 3 4 5	11' 21'' 22'' 13' 23'' 16' 25'' 14' 26''	3.31(2) 3.46(2) 3.58(3) 3.55(2) 3.24(3) 3.47(1) 3.49(2) 3.54(2) 3.51(2) 3.68(2)	1 2 3 4 5	19' 29 24'' 14 28' 13' 23' 18 21' 17''	3.66(5) 3.70(4) * 3.74(9) 3.59(4) * 3.61(7) 3.64(7) 3.65(4) 3.69(5) * 3.62(7)	14225336	4** 7** 5** 8* 8** 6** 9**	4.05(1) 3.57(1) 3.77(1) 3.55(1) 4.07(1) 3.71(1) 3.65(1) 3.69(1) 3.77(1)	3.99(5) 3.68(5) 3.79(5) 3.65(5) 4.01(5) 3.77(5) 3.68(5) 3.76(5) 3.78(5)
. 6	15'	3.42(3)		27"	3.46(7)	Ave	rage	3.76	3.79
7 8 9	18' 29'' 19' 27'' 17' 28''	3.54(4) 3.50(3) 3.52(3) 3.58(2) 3.58(3) 3.58(2) 3.58(2) 3.56(3)	6 7 8 9	17 28'' 16' 26' 25 26 15' 25'	3.60(7) 3.70(4) * 3.74(7) * 3.58(5) 3.70(5) 3.72(3) * 3.66(3) * 3.53(5) 3.73(4)	*These distances are not considered to be hydrogen bonds and are not included in the average.			
Average 3.51		Ave	rage	3.64					

hydrogen bonds. Thus we have in selenourea a system in which the hydrogen bonds are so weak that some have been sacrificed in favour of compactness, that is, to gain energy by van der Waals interaction. The intrachain hydrogen bond lengths are listed in Table III-3A, and the short interchain N...Se distances are listed in Table III-3B, together with their interpretation in terms of hydrogen bonding.

Table III-3C contains the Se...Se contact distances, both at low and room temperatures. The average values are 3.79 Å at room temperature and 3.76 Å at -100°C. According to Pauling (1960) the van der Waals radius of selenium is 2.0Å. However, he chose this value in the nineteen thirties on the basis of a few structures, and recent work consistently suggests a smaller value. The distribution of 51 Se...Se distances all determined at room temperature cited in the literature <sup>\*</sup> confirm a value of about 1.9Å. The shortening of these distances with temperature is of the order of magnitude to be expected for van der Waals contacts (see section on thermal expansion), and illustrates the care required in comparing intermolecular contacts at different temperatures.

Ketelaar, Hart, Moerel and Polder (1939), Burbank (1951), Marsh (1952), Marsh, Pauling and McCullough (1953), Tsernoglou (1966), Barnighausen, von Volkmann and Jander (1966).

#### The three molecule chain in the urea series.

The three molecule hydrogen bonded chain occurs both in the inclusion compounds and in selenourea. The chains are so similar that the increase in c-axis length in the series is a direct measure of the increasing diameter of the chalcogen. However, the situation is complicated because direction of tilt of the molecules in the thiourea adduct is uncertain, although the angle involved is well determined, being about ten degrees. This arises because the sense of this tilt only has an effect on the  $l \neq 0 \mod 3$  structure factors, and even then the contribution is small. Lenné chose the direction to be opposite to that in the urea compounds, where the angle of tilt is about seven degrees, on the basis of slightly better agreement for the hkl layer line. Since the sense of tilt found in the urea compounds recurs in selenourea, where the corresponding angle averages seventeen degrees, this choice now forms a stumbling block to a comprehensive structural description of these compounds, which I shall nevertheless attempt to present.

When we attempt to draw further analogies between these structures, we find the mode of cross-linking of the chains differs for each compound. Figure III-7 is based on part of Lenné's paper, and represents a c/3 deep slice of the urea- and thioureahydrocarbon complexes projected on the (001) plane. Each spiral





The crystal structures of the urea- and thiourea- hydrocarbon complexes.





The crystal structure of selenourea, idealised for comparison with Figure III-7.

contains a three-fold screw axis parallel to <u>c</u>. The urea and thiourea molecules form the honeycomb arrangement described previously and the hydrocarbon molecules, the honey, are left out for clarity. In these crystals the between chain hydrogen bonding from any molecule is to two oxygen (or sulphur) atoms on different molecules of the same adjacent chain. Alternate chains point in opposite directions in the (OOL) projection. The basic difference between the two inclusion compounds is that in the thiourea case alternate chains also have the opposite sense of screw, which imposes a centre of symmetry and a rhombohedral centering on the lattice.

If we compare Figure III-8, which represents a similar slice of the selenourea structure, we find that the individual chains have been rotated, so that all nine have an identical orientation in the (OOI) plane and differ only in z co-ordinate Thus all nine chains in selenourea stand in the same relationship to one another, and we cannot describe the structure by assigning a specific chain to the basic structure or to the channel. A further illustration of this rearrangement of the chain orientation in the (OOI) plane is provided by Figure III-9, where the a and c axis lengths for the three crystals are compared to the van der Waals radii (Pauling, 1960) of the corresponding chalcogens. This shows clearly that the increase in c axis length results simply from the greater chalcogen diameter. However the a axis lengths only



The axis lengths in the spiral chain compounds compared to the corresponding chalcogen radii.

follow this trend for the inclusion compounds, whereas selenourea has a markedly shorter (by about one Angstrom) a-axis than would have been expected based on an 18 molecule honeycomb structure. It therefore appears that the axial ratio is a reflection of the orientation of the chains in the (OOl) plane for all these compounds. In this connection it is interesting to note the report (Mima, 1959) that the thiourea/8-oxocamphor adduct  $(C_{10}H_{14}O_2 \cdot 3SC(NH_2)_2)$  has an axial ratio of 0.81, compared to the usual value of 0.774. This suggests the possibility of rotation of the chains, to allow hydrogen bonding to the oxygen atoms of the terpene.

Another aspect of the propensity of the urea series to form these three molecule spiral chains is the high solubility of thiourea in the selenourea structure. Crystallisation of a solution of both compounds in water yielded mixed crystals with the selenourea structure. These showed diffuse spots, near the Bragg reflections, which corresponded to the pure selenourea spacing (see Wooster, 1964). The Bragg reflections themselves corresponded to an apparently isotropic reduction in lattice parameter of one per cent, or a volume per molecule of  $94.58^3$ . Since the corresponding values for the pure components are  $90.38^3$  (thiourea) and  $97.48^3$ (selenourea), this mixed crystal contains a large mole fraction, about 0.4, thiourea.

We can also calculate, considering the molecules to be hard', the volume of the voids in the crystal structure, and hence

the efficiency of the packing in the crystal. Using the Kitaigorodskii (1960) formula, we can show that the molecular volume increase in going from thiourea to selenourea is  $5.7^{3.5}$ . Since packing coefficients for molecular crystals are commonly about 0.65, we expect a crystal volume increase of about  $9^{3.5}$  (5.7/0.65). That the actual increase is less than this reflects the more efficient packing in the selenourea structure.

This expected increase of about  $9^{A^3}$  is close to that reported by Dvoryankin and Ruchkin (1962) for the orthorhombic modification (volume per molecule  $99.7^{A^3}$ ). If this thioureatype phase exists for selenourea, then, if we accept Kitaigorodskii's arguments, we can predict it to be metastable with respect to the trigonal form.

#### Twinning in selenourea.

The relative z co-ordinates of the various chains in selenourea are indicated in Figure III-10, using the nomenclature of Table III-2. Figure III-10 also contains a similar description of the twin configuration, which can be considered as the same enantiomorph of selenourea rotated about [100] so that <u>c</u> becomes -<u>c</u>. (No attempt was made in this study to distinguish between the enantiomorphs represented by the space groups P3<sub>1</sub> and P3<sub>2</sub>, which probably occur equally frequently. Instead P3<sub>1</sub> was assumed

Figure III-10

DB E C/ C = C = C= D В

A hypothetical switching mechanism for selenourea, showing the structural relationship between the twins. The letters identify the individual chains according to the scheme of Table III-2. throughout.) One of several possible transformations from one twin to the other are indicated, involving movement of two chains by about 1.8%. This distance is large compared to that involved in known ferroelectrics, and such a movement of chains would require a gross change in the hydrogen bonding arrangement, so ferroelectricity is precluded. However, the similarity described encourages the production of such twins during the growth of crystals.

#### Thermal expansion of selenourea.

Thermal expansion coefficients for molecular crystals are of interest, in view of their relationship to the detailed crystal structure, particularly the strength and direction of interatomic interactions (Lonsdale, 1959). The magnitude of the thermal expansion coefficient  $\alpha_x$  is related to the extent of anharmonicity of lattice vibrations propagating in the direction  $\underline{x}$ , which in turn depends on the strength of the interatomic interaction, whether covalent bond, hydrogen bond or van der Waals forces, and increases in the order just outlined. Some examples of the magnitudes involved are listed in Table III-4, together with the selenourea values.

Of particular interest to the present study are the thermal expansion coefficients for urea (Gilbert, 1955), the structure of

# TABLE III-4.

Some typical coefficients of thermal expansion.

Substance	Temperature Range ( <sup>°</sup> C)	<b>∝</b> x10 <sup>6</sup> /°c	Bond Type
Diamond	23 55	1	Covalent.
Ice	-12090	∝ <sub>a</sub> 52 ∝ <sub>c</sub> 46	Hydrogen bonds.
Benzene	-1933	∝ <sub>a</sub> 119 ∝ <sub>b</sub> 106 ∝ <sub>c</sub> 221	van der Waals.
Urea	-183 20	∝ <sub>a</sub> 76 ∝ <sub>c</sub> 24	Hydrogen <sup>.</sup> bonds.
Selenourea	-100 20	∝ <sub>a</sub> 46 ∝ <sub>c</sub> 36	Hydrogen bonds.

which (see Introduction) is characterised by a fairly open arrangement of hydrogen bonds. They result from a large librational amplitude about the C-O bond axis, along with an increased vibration of all the atoms of the molecule perpendicular to its plane. This latter effect is found in general for planar molecules; in crystalline benzene, for example, the expansion in the <u>c</u> direction (almost perpendicular to the molecular plane) is about twice its value at right angles. Gilbert's conclusions were confirmed by the diffuse X-ray scattering for urea (Gilbert and Lonsdale, 1956). The nuclear magnetic resonance study also indicated motion about the C-O axis.

The thermal expansion co-efficients for selenourea are within the range common for small hydrogen bonded molecules (ice, urea, for example). However the interpretation here is complicated by the existence of Se...Se van der Waals contacts which are nearly perpendicular to  $\underline{c}$  and which expand by one per cent over the temperature range used. This, coupled with the fact that the normals to the molecular planes are also almost perpendicular to  $\underline{c}$ , results in  $\boldsymbol{\alpha}_{a}$  being somewhat greater than  $\boldsymbol{\alpha}_{c}$ , and comparison of interatomic distances at the two temperatures suggest that their relative increase does follow the predicted order: van der Waals contacts, hydrogen bonds, covalent bonds, with insignificant changes in the last. The insensitivity of covalent bond lengths to thermal expansion has been found by all workers in this field, a particularly careful study being that of Lonsdale, Milledge and El Sayed (1966) for

anthraquinone, where the structure was determined at five temperatures in the range -170°C to 20.5°C. Sklar, Senko and Post (1961) showed that the same was also true of urea, and that the expansion in that case simply resulted in increases in the hydrogen bond lengths.

#### Refractive index of selenourea.

Bunn (1945) points out that the anisotropic form of index of refraction can be related to crystal structure in the following way. For chain molecules the highest index will be along the chain direction, and for planar molecules, the present case, the lowest refractive index will be perpendicular to the molecular plane. Knowing the structure of selenourea this would predict  $n_c$   $n_a$ . This is in fact the case, for the necessary measurements have been performed by Kondrashev and Andreeva, who simply quote the indices as larger and smaller,  $n_g = 1.84$ ,  $n_p = 1.79$ . However their calculated value of the molar refractivity identifies these as  $n_c = n_g$ ,  $n_a = n_p$ , since  $\overline{n} = (n_c n_a^{-2})^{1/3}$  for trigonal crystals.

#### Conclusion.

The crystal structure of selenourea contains nine independent but identical molecules, the geometry of which is consistent with

1 1 1 1 1 m

the analogues urea and thiourea, and with the substituted selenoureas so far studied. The structure confirms the existence of N-H...Se hydrogen bonds. It also suggests that in the urea series a three-fold spiral chain is a particularly stable configuration. However, perhaps the most intriguing problem to the crystal chemist - the rationale behind the large number of nonequivalent molecules - has also proved to have perhaps the most interesting solution. The directional hydrogen bonding and the non-directional van der Waals forces in selenourea reached a complex compromise which sheds light on the competing roles of these effects in molecular crystal structures.

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